

Highly Transparent and Conductive Electrodes Enabled by Scalable Printing-and-Sintering of Silver Nanowires

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Abstract. Silver nanowires (Ag NWs) have promised well for flexible and transparent electronics. However, It remains an open question on how to achieve large-scale printing of Ag NWs with high optical transparency, electrical conductivity, and mechanical durability for practical applications, though extensive research for more than one decade. In this work, we propose a possible solution that integrates screen printing of Ag NWs with flash-light sintering (FLS). We demonstrate that the use of low-concentration, screen-printable Ag NW ink enables large-area and high-resolution patterning of Ag NWs. A critical advantage comes from the FLS process that allows low-temperature processing, short operational time, and high output rate – characteristics that fit the scalable manufacturing. Importantly, we show that the resultant Ag NW patterns feature low sheet resistance (1.1-9.2 Ohm/sq), high transparency (75.2-92.6%), and thus a remarkable figure of merit comparable to state of the art. These outstanding properties of Ag NW patterns, together with their scalable fabrication method we proposed, would facilitate many Ag NW-based

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3 applications, such as transparent heater, stretchable displays, and wearable devices; here, we demonstrate
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5 the novel design of flexible and transparent radio frequency 5G antennas.
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8 **Keywords:** Screen Printing, Silver Nanowires, Flash Light Sintering, Transparent Electronics
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11 12 13 14 15 16 17 **Introduction**

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20 With the recent interest in optically transparent electronics, transparent conductive electrodes have been
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22 extensively studied[1, 9, 16, 49, 50, 60]. In particular, various emerging nanomaterials, including
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24 conductive polymers[61], carbon materials (carbon nanotube[23], graphene[42]), metal nanowires[7, 12,
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26 45], and metal mesh[30, 51], have been leveraged for the development of electrodes with high transparency
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28 and conductivity. Among these materials, silver nanowires (Ag NWs) are quite promising as many reports
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30 revealed that the Ag NWs could achieve thin films with sheet resistance and optical transmittance
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32 comparable to conventional transparent metal oxides[33, 46]. Another mechanical advantage of Ag NWs
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34 is their mechanical flexibility that may maintain electrical conductivity even under external deformation[15,
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43 A number of issues need to be addressed before Ag NWs can truly contribute to transparent electronics.
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45 One is to pattern Ag NWs in deterministic shapes and sizes. A variety of techniques have been utilized to
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47 pattern the Ag NWs, such as photolithography[48, 51], shadow masking[5, 54], ultraviolet ozone surface
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49 treatment[27, 58], and dry transfer process[38]. However, these techniques either require multiple
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51 processing steps or suffer from limited patterning size that also limits the patterning efficiency. One-step
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53 printing seems to be an alternative method that allows efficient, scalable manufacturing. However, inkjet
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55 printing of Ag NWs requires short nanowires that can pass through the nozzles without clogging; Screen-
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57 printable and gravure-printable inks need either high Ag NWs loading[2, 19, 37, 55] or added insulating
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binders[14]. As a consequence, the printed electrodes typically compromise in functional performances, including low conductivity ($<10^6$ S/m)[18][10] and poor optical transparency ($<75\%$)[3, 14, 35]. It, therefore, remains an open question on how to achieve large-scale printing of Ag NWs with high optical transparency, electrical conductivity, and mechanical durability for practical applications, though extensive research efforts in this area.

In this work, we developed a custom Ag NWs ink suitable for screen printing. With the help of screen printing and xenon flash-light sintering (FLS), we fabricated large-area electrodes with high conductivity and optical transparency simultaneously. The screen-printable Ag NW ink with low Ag loading (about 0.9 wt.%) is developed by introducing polyvinyl pyrrolidone (PVP) with ultrahigh molecular weight ($M_w = 2 \times 10^6 - 3 \times 10^6$ g/mol) as a binder material, which is beneficial to highly transparent patterns. The ink composition is optimized to allow the deposition of Ag NW patterns with a good spatial resolution (approximately 50 μm for line width and line spacing). Then, FLS is utilized to sinter the printed Ag NW patterns in milliseconds, enabling more than 70% enhancement of electrical conductivity (as compared to non-sintered electrodes) without any degradation in the optical transparency. We obtain scalable Ag NW patterns with extremely low sheet resistance (1.1-9.2 Ohm/sq), ultrahigh transparency (75.2-92.6% in transmittance), and superior figure of merit (FoM, more than 1100). Moreover, the printed and flash-sintered Ag NW patterns demonstrate outstanding mechanical flexibility, showing negligible changes in resistance after 1000 bending cycles. These features fit flexible, transparent, and high-performance electronic devices. As a proof of concept, we demonstrate a multi-frequency radio frequency (RF) antenna, capable of working in the WiFi, Bluetooth, and 5G bands.

Experimental Section

Raw Materials

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3 Silver nitrate (AgNO_3), ethylene glycol (EG), iron (III) chloride (FeCl_3), polyvinyl pyrrolidone (PVP, M_w :
4 55,000, 1,300,000), propylene glycol, isopropanol and 2-butoxyethanol are purchased from Sigma-Aldrich.
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7 PVP (K120) is obtained from Ashland.
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10 **Synthesis of Ag NWs**

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13 The Ag NWs are synthesized through a modified polyol process. Briefly, PVP ($M_w=55,000$, 0.85 g,
14 $M_w=360,000$, 0.85 g) are added to EG (200 mL) with stirring. Next, AgNO_3 (2 g) is added to the PVP
15 solution. After complete dissolution, FeCl_3 solution (28 g, 0.6 mM in EG) is added and stirred for 2 minutes.
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18 Then, the mixture is transferred to a preheated round-bottom flask at 130 °C and allowed reaction for 4 h.
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21 After cooling to room temperature, the Ag NWs are washed using acetone and ethanol for 3 cycles.
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26 **Formulation of Ag NWs-Based Screen-Printable Ink**

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29 The ink is formulated by mixing the washed Ag NWs with PVP solution. First, 3 g of PVP (K120, $M_w=2 \times$
30 $10^6 - 3 \times 10^6$ g/mol) is dissolved in a solvent mixture containing of propylene glycol, isopropanol and 2-
31 butoxyethanol with a ratio of 8:1:1. Then, Ag NWs are mixed with PVP solution and agitated thoroughly
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34 using a propeller stirrer at a speed of 1500 rpm for 20 min to obtain the Ag NW ink.
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39 **Screen Printing of Ag NWs**

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42 The Ag NWs patterns and antennas are printed on clean PET substrates using a screen stencil with a mesh
43 count of 325, a wire diameter of 20 μm , and an emulsion thickness of 10 μm . Next, the printed samples are
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46 dried at 80 °C for 10 min in an oven to evaporate solvents. Then, the Ag NW samples are immersed in
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49 warm water (60 °C) or ethanol for 5 min to remove most of the PVP (k120), while remaining the Ag NWs
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52 at the original position.
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55 **Flash Light Sintering**

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Flash light sintering is conducted using Novacentrix Pulse Forge 1300 system. During processing, the distance between the sample and flash lamp is kept at 5 mm. The pulse length, pulse voltage and pulse number are optimized as given in the text.

Antenna simulation

High Frequency Structure Simulator (HFSS) is utilized to perform the simulation for the fractal antenna. A 3D model, containing two conductor layers and a dielectric layer, is constructed. The dielectric substrate has a dielectric constant of 2.36, a dielectric loss tangent of 0.005 at 1 GHz, and a thickness of 0.1 mm. The conductor has an electrical conductivity of 5×10^6 S/m and a thickness of 1 μ m. A discrete 50-ohm port is modeled as the source.

Characterization

The morphologies of Ag NWs are characterized by SEM (ZEISS Merlin). The structure of Ag NWs is characterized by XRD (Bruker D2 PHASER). Thermal gravimetric analysis is performed using a TG 209 F1. The rheological behaviour of the formulated Ag NW ink is tested using an AR1500 rheometer (TA Instruments). The sheet resistance of the Ag NW patterns is measured using a four-probe resistance tester (CMT-SR2000N). The transparency is evaluated using a UV-Visible Spectroscopy (Thermo Evolution 600). The thickness of the Ag NW film is measured using a surface profiler (Veeco Dektak 150). RF measurements are performed using an Agilent N5225A Vector Network Analyzer. The S-parameters of the antennas are acquired from 1 to 6 GHz after RF calibration conducted based on the short-open-load (SOL) method to eliminate parasitic effects of the measurement system. The antenna gain and radiation pattern of the antenna are obtained in a tapered anechoic chamber (Satimo Starlab). Before measurement, the chamber is calibrated using NSI 2 - 18 GHz horn antenna.

Results and Discussion

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3 To ensure a stable and high resolution (i.e. 50 μm) printing of Ag NWs, we introduce a viscous polymer
4 matrix containing PVP and a solvent mixture (propylene glycol, isopropanol, and 2-butoxyethanol) to
5 disperse high-aspect-ratio Ag NWs (length-to-diameter ratio > 600 ; Fig. S1) with low Ag loading (about
6 0.9 wt.%; Fig. S2a). The prepared Ag NW ink is stable, and no settlement of Ag NWs is observed after two
7 months at 4 $^{\circ}\text{C}$ (Fig. S2b). A shear-thinning behaviour is observed in the rheological curve, and the ink had
8 a viscosity of 4.7 Pa s at a shear rate of 1 s^{-1} (Fig. S2b). With this ink, desired patterns with high transparency
9 are precisely deposited onto the target substrates at speed higher than 200 mm/s. When the printing process
10 is complete, the printed Ag NW patterns are cured at 80 $^{\circ}\text{C}$ for 5 min to evaporate the solvents and obtain
11 dry and dense Ag NWs films. Then, a warm water bath (60 $^{\circ}\text{C}$) or ethanol bath, is used to help the Ag NW
12 patterns recover their conductivity via PVP dissolving in the bath solution[19, 37], as can be observed in
13 the SEM images of the Ag NWs before and after the bath in Fig. S3. Following the binder removal step,
14 the conductive Ag NW patterns are exposed to pulsed flash-light to weld the NW junctions to further
15 enhance the electrical conductivity.
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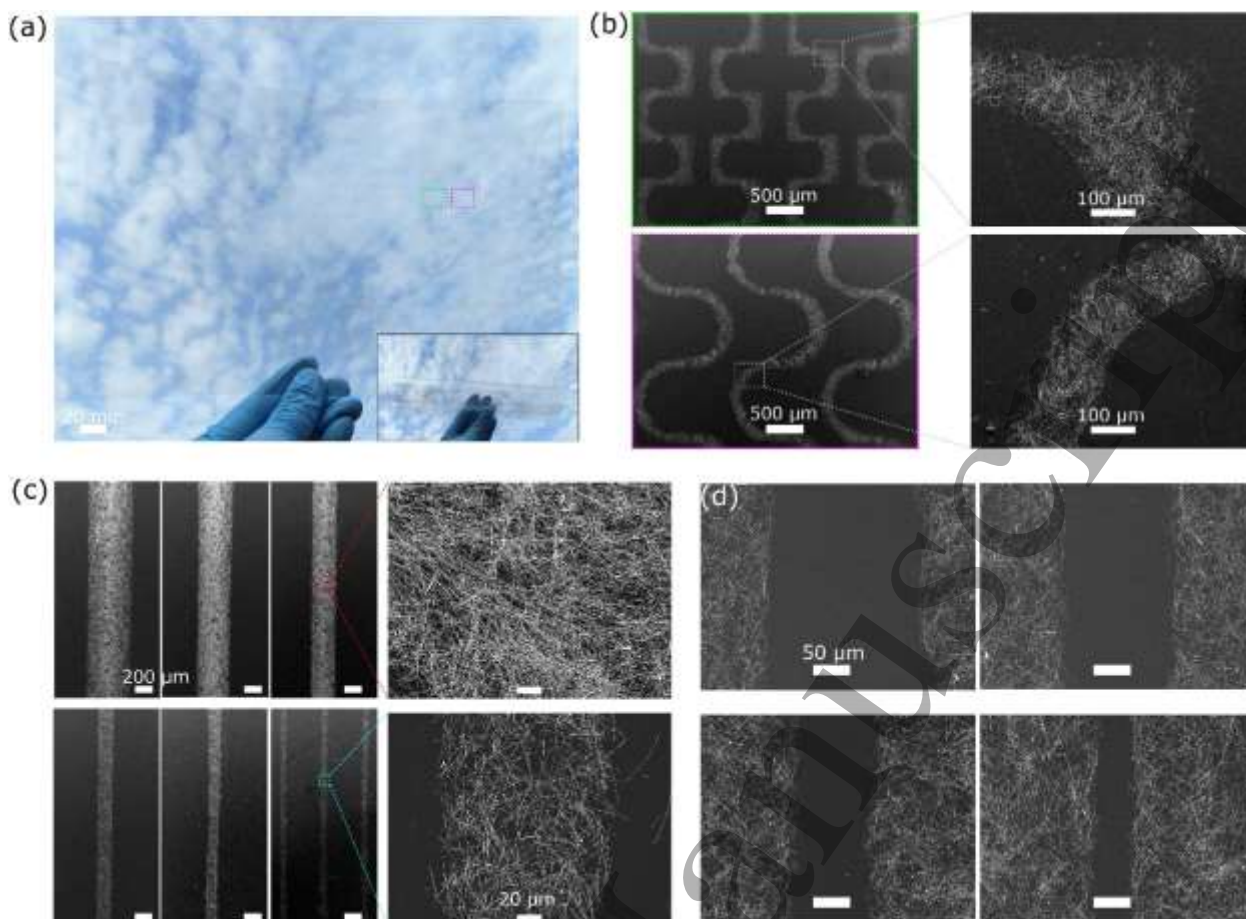


Fig. 1. Large-area and high-resolution printing of Ag NWs: (a) digital photograph of the screen-printed Ag NW patterns with an optical transmittance of 92% at $\lambda = 550$ nm on a PET substrate. (inset: the rolling status of the Ag NW patterns, demonstrating the flexibility); (b) SEM images of the screen-printed Ag NW serpentine line patterns and spiral line patterns (the right panel displays magnified SEM images of the edge of the Ag NW patterns); (c) SEM images of the screen-printed Ag NW lines with different line widths from 500 μm to 50 μm (the right panels present the Ag NW network and the edge of the Ag NW line); (d) SEM images of the screen-printed Ag NW patterns with different spacing of 200, 150, 100, and 50 μm .

The developed Ag NW ink and the screen-printing technique enabled one-step, direct patterning of Ag NWs in a large area and in high resolution. Fig. 1a and Fig. S4 present digital photographs of the printed Ag NWs patterns with different transparency (transmittance at $\lambda=500$ nm: from 92% to 82% and 75%) on flexible polyethylene terephthalate (PET) substrates with a dimension of 200×200 mm^2 , and on a rigid glass substrate with a diameter of 100 mm. From the scanning electron microscope (SEM) images in Fig.

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3 1b, we found the precise, sharp, and smooth edges of the screen-printed Ag NW patterns. In addition,
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5 printed Ag NW lines with various line width are achieved successfully, and the narrowest is approximately
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7 50 μm , as shown in the SEM images in Fig. 1c. Note that, line width of about 50 μm is close to the limit
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9 for screen-printing technique due to the resolution of the screen, and this has been confirmed by many other
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11 reports in the literature[20, 37]. In addition to the printed fine lines, different line spacing is attained. As
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13 displayed in Fig. 1d, all the printed spacing are in uniform shapes and smooth edges, and the printed spacing
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15 are 206(\pm 10), 153(\pm 6), 102(\pm 9), and 48(\pm 5) μm .
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20 The morphological features of the pristine and the flash-sintered Ag NWs are investigated via SEM. As
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22 Fig. S5a depicts, the pristine NWs have loose contact points with a gap between the NWs, resulting in high
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24 electrical resistance[40]. However, the flash-sintered Ag NWs demonstrate strong welded junctions, and
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26 the gaps between NWs become smaller and even disappear, as Fig. 2a, 2b and Fig. S5b illustrate. The
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28 welded junctions decrease the resistance of the Ag NWs patterns and hence improve the electrical
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30 conductivity.
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35 When placed under the flash lamp at a distance of 5 mm, the Ag NWs are exposed and illuminated by a
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37 high-intensity pulsed light with a broad wavelength of 200–1500 nm. It is believed that such illumination
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39 will generate highly localized heating only at the NW junctions[6, 29, 44]. This phenomenon, also known
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41 as “hot spot”, can result in the welding of the Ag NW junctions because of the high-temperature thermal
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43 process. To further understand the effect of the FLS on lowering the resistance of the screen-printed Ag
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45 NW patterns, we study various parameters of the sintering process, including pulse length, pulse voltage,
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47 energy density (which depends on the pulse length and applied voltage) and pulse number. We choose Ag
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49 NW patterns with an initial sheet resistance of approximately 46 ohm/sq and a transmittance of 91.6 % at
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51 $\lambda = 550$ nm as a reference sample. First, the sample is sintered with different pulse lengths (corresponding
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53 to different energy densities), keeping the same pulse voltage of 350 V and pulse number of 40. The
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55 measured sheet resistances decrease almost linearly from 46 ohm/sq to 11 ohm/sq as the pulse length
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3 increases from 100 μs to 600 μs (corresponding to 0.93 J/cm^2 , Fig. 2c). Then, the resistance remains nearly
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5 constant (about 10.3 ohm/sq) when the pulse length is further increased to 900 μs in which energy density
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7 corresponds to 1.385 J/cm^2 , indicating the self-limiting behaviour of the inter-NW junctions. As expected,
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9 the transmittances of the Ag NW electrodes remain nearly constant (about 91.5%) during the flash sintering
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11 process even when exposed to high light energy density of 1.385 J/cm^2 [6, 53]. Next, as the pulse voltage
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13 increases from 100 to 350 V using 50 V step (pulse length: 600 μs , pulse number: 40), the sheet resistances
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15 decrease from 45.9 ohm/sq to 10.1 ohm/sq (Fig. S6a) and reach a value of 10.3 ohm/sq for pulse voltages
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17 >400 V. Finally, when the sample is sintered for 40 pulses (24 ms total), the sheet resistance dramatically
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19 decreases from 46.6 ohm/sq to 9.6 ohm/sq (Fig. S6b). When further increasing the pulse numbers (over 50,
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21 corresponding to a exposure time of 30 ms), the sheet resistance only slightly increases to 10.4 ohm/sq [11].
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23 To prove the effectiveness of the flash-sintering process, we print Ag NW patterns with various sheet
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25 resistance (5.8-42 ohm/sq) and transmittance (72-92.2%) via tuning the printing passes, and then sinter
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27 under FL with optimized parameters (pulse length: 600 μs ; pulse voltage: 375 V, pulse number: 40). As
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29 presented in Fig. S7, the sheet resistances of all the screen-printed Ag NW patterns dramatically decrease
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31 by more than 70% (the maximum is 82.7%; Fig. S7) after exposure to FL for a total duration of 24 ms. To
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33 quantitatively demonstrate the superior electrical and optical performance of our screen-printed, flash-
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35 sintered Ag NW patterns and compare with the transparent electrodes in literature, we introduce the figure
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37 of merit (FoM) to define the trade-off between the electrical conductivity and optical transparency by
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39 considering the sheet resistance (R_s) and transmittance (T , at $\lambda=550$ nm) as expressed in Equation (1):
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$$FoM = \frac{188.5}{R_s (T^{-1/2} - 1)} \quad (1)$$

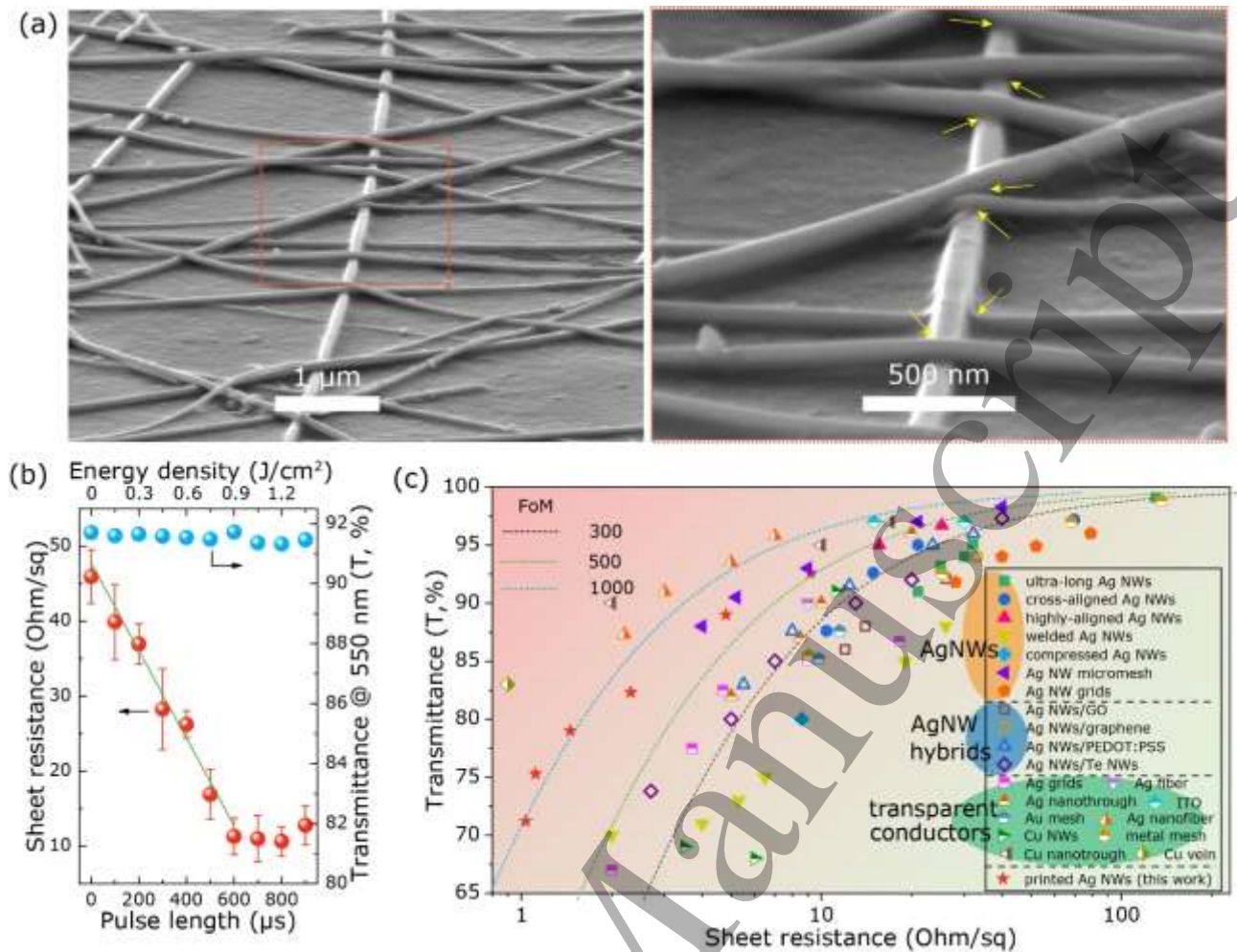


Fig. 2. (a) Tilted view SEM image of Ag NWs after the FLS process. The right panel is an enlarged SEM image in higher magnification, and the yellow arrows indicate the welded junctions between NWs; (b) the sheet resistance and optical transmittance at $\lambda = 550$ nm of the printed Ag NW patterns under different pulse lengths and energy densities of the flash light; (c) the sheet resistance versus optical transmittance (at $\lambda = 550$ nm) for the screen-printed, flash-sintered Ag NW patterns, along with some selected, important transparent electrodes reported in the literature (the dotted lines represent FoM of 300, 500, and 1000 according to Equation [1]).

Evidently, lower sheet resistance and higher transmittance values result in high FoM, which is highly desired for optoelectronic applications. Fig. 2c shows three fitted curves based on Equation (1) with FoM values of 300, 500, and 1000 (dashed lines). The transmittance of our screen-printed, flash-sintered Ag NW patterns with various sheet resistances are also indicated. All the FoM values are located on the left side of $\text{FoM} = 300$ at high transmittance (larger than 90%), indicating that the FoM values are over 300. In

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3 comparison, most of the previously reported high-quality transparent conductive films exhibit FoM values
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5 below 300, with only some exceeding 500. Our printed Ag NW films exhibit the lowest sheet resistance
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7 with relatively high transmittance in comparison with values reported previously in the literature for Ag
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9 NWs-based[4, 26, 29, 33, 47, 51, 52] and Ag NW hybrid[31, 36, 39, 57] electrodes. In addition, the highest
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11 FoM value measured for our printed Ag NWs electrodes exceeds 1100, further demonstrating the
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13 superiority over other transparent conductor technologies, such as Ag grids[22], Ag nanotrough[50], Ag
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15 fiber[1], metal mesh[32], copper (Cu) NWs[17], gold (Au) mesh[21] and commercial indium tin oxide[8].
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17 Only a handful of studies on Ag nanofiber[43], Cu nanotrough[50] and Cu veins[59], reported higher FoM
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19 than our printed Ag NWs films. However, our method allows for direct, high throughput printing of Ag
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21 NWs electrodes of arbitrary shape over a large area, temperature-sensitive substrates.
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27 After the FLS, the electrical conductivity (σ) of the Ag NW patterns has been calculated based on the
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29 measured resistance (R) and the line geometries (i.e., line width [w], line length [l], line thickness [t]),
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31 following the Equation: $\sigma = l/(Rwt)$. As illustrated in Fig. 3a, the measured resistance per unit length
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33 decreases sharply when the printed line width increases from 50 μm to 300 μm , followed by a less
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35 pronounced drop for line widths $>300 \mu\text{m}$. This is because that some Ag NWs did not pass through the
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37 opening mesh to form lines with a width of about 50 μm due to the smaller size of the actual opening mesh
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39 (less than 38 μm) than the Ag NWs (20-60 μm). When the line width increases, the amount of Ag NWs
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41 transferred to the substrates increases, consequently leading to the formation of thicker electrodes. The
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43 thickness of the Ag NW electrodes is found to increase from 106 to 206 nm as the line width increased
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45 from 50 μm to 300 μm , followed by a smaller increase for line widths in the range of 300 to 500 μm (Fig.
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47 3b). The result is consistent with the calculated conductivity of the printed lines, where the conductivity
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49 increases from 2.1×10^6 to 5.5×10^6 S/m as the line width increases from 50 to 500 μm (Fig. 3c). The
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51 calculated conductivity of our screen-printed, flash-sintered Ag NW electrodes is highly competitive to
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53 values found in the literature Ag NW electrodes processed via different methods[19, 37].
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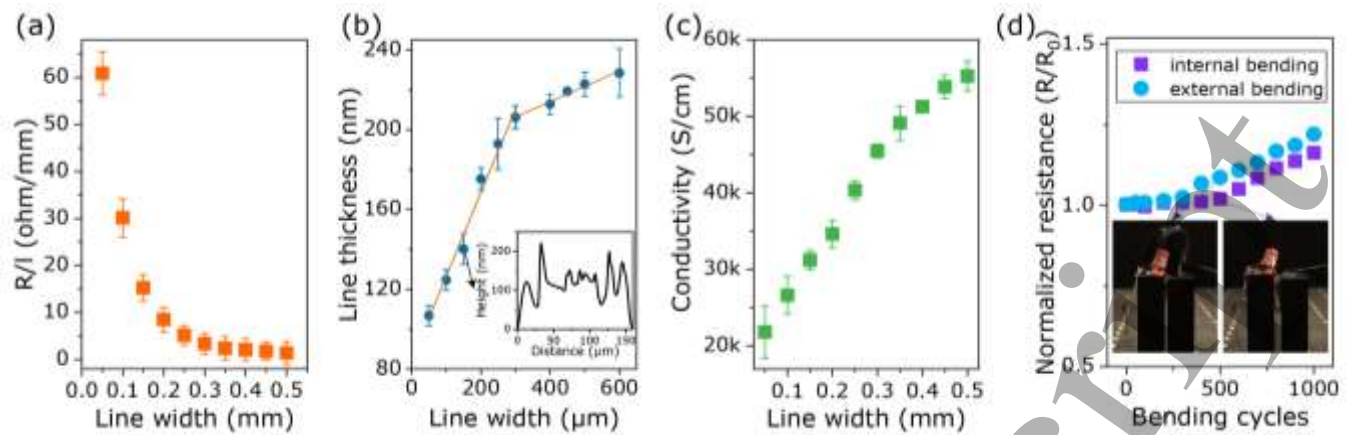


Fig. 3. (a) Resistance per unit length of the Ag NW lines as a function of line width. (b) Line thickness of the Ag NW lines as a function of line width. The orange line is a fitting curve. Inset is profile height of a printed Ag NW line with a width of 150 μm . (c) The calculated conductivity of the printed and sintered Ag NW patterns as a function of line width from 0.05 mm to 0.5 mm. (d) Normalized resistance change of Ag NW lines under cyclic bending. Insets are LED light connected to the Ag NW line.

In addition to the outstanding electrical and optical characteristics, mechanical stability is also important factor for practical applications in flexible and transparent electronics. To investigate the reliability of the screen-printed Ag NWs patterns under mechanical stress, resistance changes of the Ag NWs are monitored during bending both in internal (compressing) and external (extending) conditions. Fig. 3d shows that after continuous bending for over 1000 cycles, the resistance of the Ag NW electrodes with a width of 0.5 mm and length of 5 mm only increases by 16% and 22% for internal and external bending, respectively. Despite this increase, the test circuit maintains its conductivity to light a LED (inset of Fig. 3d), without evidence for any catastrophic failure/fracture during bending cycles.

The high optical transparency, excellent electrical conductivity, and outstanding mechanical stability make our Ag NW electrodes ideal for application in various electronic devices. To demonstrate the potential of the technology, we apply the screen-printed, flash-sintered Ag NW electrodes in RF field. In particular, we design and characterize a flexible, transparent fractal antenna for Wi-Fi (2.4 GHz), Bluetooth (2.4 GHz), and low-band 5G (3.5-4.2 GHz) applications. Fig. S8 provides the detailed geometric dimensions and 3D

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3 model of the antenna in the full-wave electromagnetic simulation software and one photo of fabricated
4 antenna. The minimum line length and the gap of the design are 0.4 and 0.15 mm, respectively. Both the
5 antenna layer and the ground layer are made of screen-printed and flash-sintered Ag NWs with sharp and
6 clear edges (magnified SEM images in Fig. 4a), demonstrating the reliability and high-resolution of our
7 screen-printing process.
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12 To quantitatively evaluate the antenna performance, we first study the antenna's reflection coefficient
13 (S_{11}), which represents the ratio of the reflected electromagnetic wave to the incident electromagnetic wave
14 for a certain frequency range. The frequency range at which the antenna has low S_{11} (below -10 dB), where
15 less than 10% of power is reflected back, is considered the impedance bandwidth of the antenna. As
16 presented in Fig. 4b, the measured S_{11} of the printed antenna matches the simulated results, with acceptable
17 discrepancies. In particular, two reflection peaks at the frequencies of 2.3 and 4.1 GHz have been observed
18 in the measurement, which is close to the simulated values of 2.3 and 4.3 GHz. The impedance bandwidth
19 are 3.3 GHz (from 1.5 to 4.8 GHz) and 2.1 GHz (from 1.9 to 3.1 GHz and from 3.8 to 4.7 GHz) for the
20 measured and simulated results, respectively, which cover the wireless standards, such as Wi-Fi, Bluetooth,
21 and low-band 5G. The antenna's radiation properties, including antenna gain and radiation pattern are
22 measured in a nearfield anechoic chamber (Star Lab from Satimo) and are depicted, along with the
23 simulated results, in Fig. 4c and Fig. S9a. Again, the trend between the simulation and measured curves is
24 similar. The differences between the measured and simulated results can be attributed to the under-
25 estimation of conductive and dielectric losses in simulations and the rough surface of the printed Ag NW
26 film. Nonetheless, the antenna demonstrates a decent impedance bandwidth and a gain of above 1 dBi for
27 the frequency range of 2.3 to 4.1 GHz. Furthermore, the antenna has a decent omnidirectional radiation
28 pattern, as expected for a monopole antenna. The E- and H-planes of the antenna radiation patterns are
29 displayed in Fig. S9a at 2.4 GHz. Overall, the performance of this low-cost printed transparent antenna is
30 comparable to standard printed circuit board-based antennas.
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3 The flexibility and reliability of the Ag NW antenna under mechanical deformations is also studied.
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5 Despite the extreme bending conditions, with a bending radius of 10 mm, no measurable change in the
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7 bandwidth is observed even after 100 bending cycles (Fig. 4d). We also investigate the optical transparency
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9 of the Ag NW antenna by examining the transmittance over the entire visible wavelength (360 nm to 800
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11 nm). The Ag NW antenna exhibits high optical transparency as displayed in the optical image in Fig. S9b,
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13 where the IMPACT logo can be clearly seen through the Ag NW antenna. The Ag NW antenna
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15 demonstrates a high transmittance of 87% at a wavelength of 550 nm, which is better than the previously
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17 reported optically transparent antennas in the literature, including metal nanoparticles (75%)[34], metal
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19 mesh (84.5%)[28], Ag coated polyester (80%)[41], conductive oxide (80%)[56], and graphene (85%)[13].
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21 To demonstrate the scalability of the process, Fig. 4e presents five printed Ag NW antennas, which are
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23 prepared under the same printing parameters, sintering process, and sealing condition. All the antennas
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25 demonstrate similar RF performance (Fig. 4f), validating the scalability, repeatability, and reliability of our
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27 fabrication procedures.
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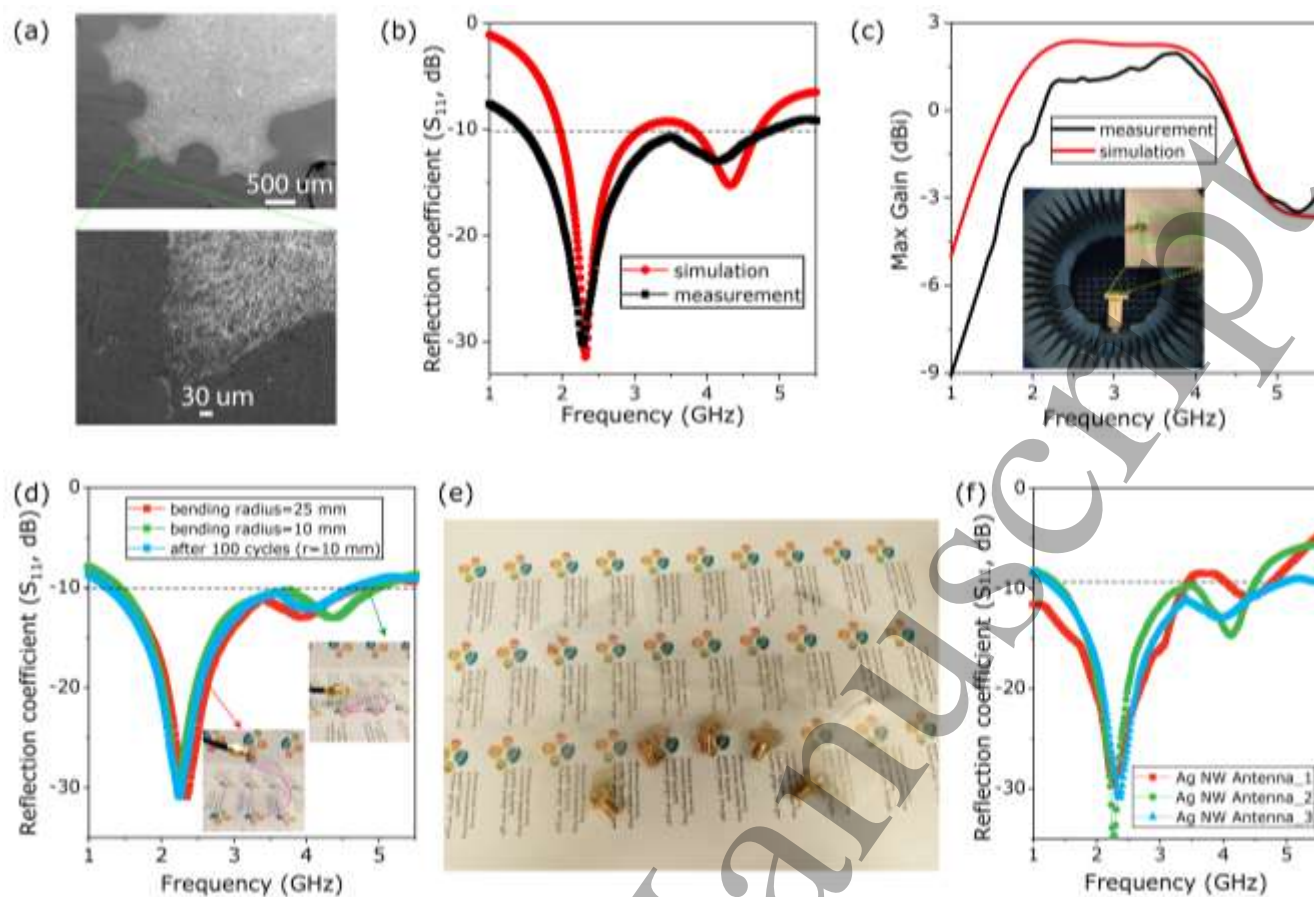


Fig. 4. RF performance of the printed Ag NW antenna: (a) SEM images of the printed Ag NW antenna (bottom panel displays the fine features of the antenna); (b) the measured and simulated return loss of the Ag NW antenna; (c) the measured and simulated antenna gain in the frequency of 1 to 5.5 GHz (inset is an Ag NW antenna under test in an anechoic chamber); (d) the measured S_{11} of the Ag NW antenna under mechanical deformation with different bending radius and cyclic bending at a bending radii of 10 mm (insets: digital photographs of the Ag NW antenna under bending conditions); (e) digital photograph and (f) the measured S_{11} of three Ag NW antennas.

Conclusions

In summary, screen-printable Ag NW ink with Ag NW loading as low as 0.9 wt% is developed via dispersing Ag NWs into a highly viscous PVP matrix. With the ink, Ag NW patterns with a large area ($200 \times 200 \text{ mm}^2$), high efficiency (up to 200 mm/s), high resolution (line width of about $50 \mu\text{m}$), high optical transparency (92.2% transmittance at $\lambda = 550 \text{ nm}$), high electrical conductivity (larger than $2 \times 10^6 \text{ S/m}$),

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3 and high FoM (more than 1100) are obtained via screen-printing and FLS. The additive manufacturing
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5 nature of screen-printing makes our Ag NW electrode technology for the mass production of flexible and
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7 transparent electronic devices. A fractal Ag NW-based antenna demonstrates well-matched RF
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9 performance with simulations, including return loss, antenna gain, and radiation pattern. The fully-printed
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11 Ag NW antennas exhibit remarkable mechanical stability under continuous bending (100 cyclic bending
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13 with a radius down to 10 mm) and high optical transparency (transmittance of 87% at $\lambda = 550$ nm). The
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15 combination of remarkable electrical, optical, and mechanical properties makes the screen-printed Ag NW
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17 electrodes a much promising technology for application in flexible, large-area, transparent electronics and
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19 RF components.
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25 **Conflicts of interest**

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28 There are no conflicts to declare.
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34
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37 Technology (KAUST).
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