

## Influence of surface properties of filtration-layer metal oxide on ceramic membrane fouling during ultrafiltration of oil/water emulsion

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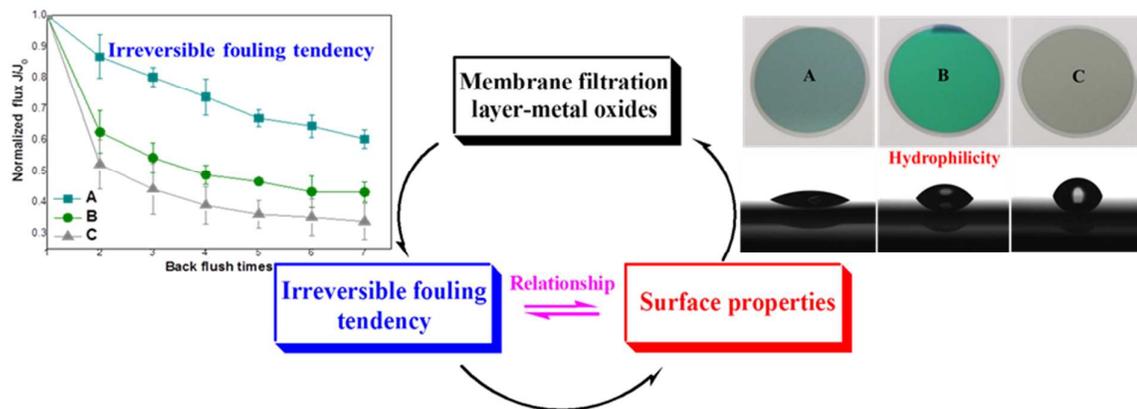
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37 **ABSTRACT**

38 In this work, ceramic ultrafiltration membranes deposited with different metal  
39 oxides (i.e., TiO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>, MnO<sub>2</sub>, CuO, and CeO<sub>2</sub>) of around 10 nm in thickness and  
40 similar roughness were tested for O/W emulsion treatment. Distinct membrane  
41 fouling tendency was observed, which closely correlated to the properties of the  
42 filtration-layer metal oxides (i.e. surface hydroxyl groups, hydrophilicity, surface  
43 charge, and adhesion energy for oil droplets). In consistent with the distinct bond  
44 strength of the surface hydroxyl groups, hydrophilicity of these common metal oxides  
45 are quite different. The differences in hydrophilicity consequently lead to different  
46 adhesion of these metal oxides towards oil droplets which consists very well with  
47 irreversible membrane fouling tendency. In addition, the surface charge of the metal  
48 oxide opposite to that of emulsion can help to alleviate irreversible membrane fouling  
49 in ultrafiltration. Highly hydrophilic Fe<sub>2</sub>O<sub>3</sub> with lowest fouling tendency could be a  
50 potential filtration-layer material for the fabrication/modification of ceramic  
51 membranes for O/W emulsion treatment. To the best of our knowledge, this is the first  
52 study clearly showing the correlations between surface properties of filtration-layer  
53 metal oxides and ceramic membrane fouling tendency by O/W emulsion.

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## 59 INTRODUCTION

60 Oily wastewater is produced in large volumes by many industries (e.g., oil and gas  
61 extraction, petroleum refining, food industry, and metal manufacturing).<sup>1,2</sup> The oil in  
62 water is commonly classified as free oil ( $D_p > 100 \mu\text{m}$ ), dispersed oil ( $D_p = 100\text{-}10$   
63  $\mu\text{m}$ ), emulsified oil ( $D_p < 10 \mu\text{m}$ ), and dissolved oil, among which O/W emulsion and  
64 dissolved oil compounds are most difficult to be effectively removed. Oil-in-water  
65 (O/W) emulsions are usually stabilized by emulsifier (e.g., surfactant) which reduces  
66 the interfacial tension between oil and water.<sup>3</sup> Most of environmental guidelines  
67 require that the maximum oil and grease concentrations in effluents should be less than  
68 10-15 mg/L (in some countries  $< 5 \text{ mg/L}$ ), making efficient removal of O/W emulsion  
69 very critical in the treatment of oily wastewater.<sup>4</sup> Most of the conventional processes  
70 for O/W emulsion treatment (e.g., gravity separation, chemical demulsification, air  
71 flotation, etc.), are not efficient enough in the treatment of stable O/W emulsions to  
72 meet the standards.<sup>5-7</sup> These processes also have disadvantage such as significant  
73 recontamination and high operation cost.<sup>8-10</sup>

74 Ultrafiltration (UF) has been widely studied for O/W emulsion treatment to replace  
75 traditional treatment because of its relatively low operation cost, steady quality of  
76 permeate, and small space requirement.<sup>11-13</sup> Ceramic membranes are considered to be  
77 superior to polymeric membranes for the treatment of oily wastewater as they have  
78 higher mechanical, chemical, and thermal stability, as well as possibly overall lower  
79 lifecycle cost.<sup>14-16</sup> However, like polymeric membranes, ceramic membranes also  
80 suffer from fouling during the treatment of O/W emulsions.<sup>17</sup> Membrane fouling

81 (including reversible and irreversible fouling) leads to severe decline of permeate flux  
82 and thus compromises the membrane performance. During the filtration of oily  
83 wastewater, reversible fouling caused mainly by the accumulation and deposition of  
84 oil foulants on membrane surface can be effectively reduced by hydraulic washing  
85 (e.g., cross flow and back flush).<sup>17, 11, 18, 19</sup> In contrast, irreversible fouling cannot be  
86 effectively reduced merely by hydraulic washing, which therefore is one of the major  
87 challenges to ceramic UF membrane in O/W emulsion treatment. The irreversible  
88 fouling can be ascribed to: 1) blockage of oil droplets inside membrane pores, and 2)  
89 strong adsorption of oil foulant on membrane surface and pores.<sup>20</sup> The first reason of  
90 irreversible-fouling is mainly related to the structure and pore size of the membrane as  
91 well as the size of oil droplets.<sup>21</sup> The second one is largely determined by the  
92 interaction between membrane filtration-layer material and oil foulant.<sup>22, 23</sup> For O/W  
93 emulsion treatment, strong oil adsorption could be the main reason causing  
94 irreversible fouling due to oil foulant properties such as high viscosity. In such  
95 circumstance, ceramic membrane filtration-layer material will play an important role  
96 for the irreversible fouling.

97 Polymeric membrane materials have been investigated intensively for their  
98 antifouling performances during oily wastewater treatment, and the influences of  
99 material properties have been widely reported.<sup>22-24</sup> In contrast, roles of ceramic  
100 membrane materials in their antifouling performances have not been systematically  
101 studied.<sup>25, 26</sup> Knowledge on relationships between surface properties of filtration-layer  
102 metal oxide and ceramic membrane fouling tendency will be critical to select

103 appropriate materials for fabrication/modification of ceramic membrane to  
104 successfully treat O/W emulsion.

105 In this research, five common metal oxides (i.e.  $\text{TiO}_2$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{MnO}_2$ ,  $\text{CuO}$ , and  $\text{CeO}_2$ )  
106 were selected for experiment due to their high stability in water and relatively low  
107 prices. These metal oxides are also widely used as supports for catalysts,<sup>27, 28</sup> thus  
108 enabling future fabrication of catalytic membranes to mitigate membrane fouling for  
109 the treatment of O/W emulsions.<sup>18, 29, 30</sup> These metal oxides were deposited separately  
110 on commercial ceramic membranes (the filtration layer is composed of  $\text{ZrO}_2$ ) with  
111 pulsed laser deposition (PLD) to achieve very thin deposition layers (around 10 nm)  
112 while without changing membrane roughness and permeability. The irreversible  
113 fouling on the deposited membranes was investigated through the treatment of crude  
114 oil/water emulsion. Surface properties of these metal oxides were comprehensively  
115 characterized in terms of surface hydroxyl groups, hydrophilicity, surface charge, and  
116 adhesion energy for oil droplets. The irreversible membrane fouling tendency was  
117 correlated to these surface properties of filtration-layer metal oxide. The relationships  
118 between surface properties of the filtration layer and the membrane fouling tendency  
119 in the treatment of O/W emulsion was, for the first time, clearly demonstrated for  
120 ceramic membrane, which can provide a basic knowledge to select/prepare/modify  
121 ceramic membranes for sustainable O/W emulsion treatment.

## 122 **EXPERIMENTAL SECTION**

### 123 **Preparation and characterization of O/W emulsion**

124 Crude oil (provided by Saudi Aramco) was used as base oil to prepare oil/water  
125 (O/W) emulsions. Sodium dodecyl benzene sulfonate (SDBS) (Sigma-Aldrich) was  
126 used as surfactant to disperse and stabilize the crude oil droplets in preparation of  
127 O/W emulsions. Details of the emulsion preparation procedure were provided in **Text**  
128 **S1** of the Supporting Information (**SI**).

129 The O/W emulsion was characterized in terms of oil-droplet size distribution and  
130 zeta potential which were analyzed with a particle size and zeta potential analyzer  
131 (Nano ZS, Malvern). Microscopic configuration of the oil droplets was characterized  
132 with confocal laser scanning microscopy (CLSM) (LSM710, Carl Zeiss) using Nile  
133 red (Sigma-Aldrich) as a stain.<sup>31</sup>

134 The emulsion has an average oil droplet size of 360 nm after 50 min sonication  
135 (**Table S1 and Figure S1, SI**). It was negatively charged with zeta potentials  
136 approaching -45.5 mV at pH 6.0. **Figure S2 (SI)** shows that the fine oil droplets  
137 distributed uniformly in the emulsion solution. During the experiments, the  
138 as-prepared emulsion was relatively stable in droplet size distribution and zeta  
139 potential.

#### 140 **Ultrafiltration of O/W emulsion with ceramic membranes**

141 TiO<sub>2</sub>, MnO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>, CuO, and CeO<sub>2</sub> (Sigma-Aldrich) were deposited separately onto  
142 the filtration layer (composed of ZrO<sub>2</sub>) of commercial ceramic membranes (50 kDa,  
143 TAMI Industries) by Pulsed-Laser Deposition (PLD) (COMPexPro201, Coherent)  
144 technique under specific conditions (**Text S2, SI**) with nearly the same coating  
145 thickness (around 10 nm). The coating of five metal oxides on membrane surface was

146 further tested by scanning transmission electron microscopy (STEM) (Titan 80-300,  
147 FEI) (**Figure S3, SI**). The elements of the coated metal oxide were confirmed with the  
148 Energy Dispersive Spectroscopy (EDS) accessory of the STEM equipment (**Figure**  
149 **S4, SI**). Morphologies of the virgin membrane and metal oxide-deposited membranes  
150 were characterized with scanning electron microscopy (SEM) (Magellan 400, FEI) and  
151 Atomic Force Microscopy (AFM) (Agilent 5500, Agilent). The surface roughness and  
152 zeta potential of these membranes were measured by AFM and Electrokinetic  
153 Analyzer for Solid Samples (SurPASS, Anton Paar), respectively. Details of the  
154 filtration procedure were provided in **Text S3 (SI)**. Irreversible fouling of the  
155 membranes were evaluated with normalized initial permeate flux of each filtration  
156 cycle ( $NIF_n$ ) which denotes the recovery extent of the permeability by hydraulic back  
157 flush. The higher NIF is, the less irreversible fouling the membrane has.

158 The normalized initial permeate flux of each cycle ( $NIF_n$ ) was calculated with  
159 equation (1).

$$160 \quad NIF_n = J_n/J_0 \times 100\% \quad (1)$$

161  $J_n$  denotes the initial permeate flux after each back flush;  $J_0$  is pure water permeate  
162 flux of the virgin membrane.

163 Instantaneous permeate flux  $J$  ( $L m^{-2} h^{-1}$ ) was calculated with equation (2).

$$164 \quad J = V/A\Delta t \quad (2)$$

165  $V$  (L) is the permeate volume;  $A$  ( $m^2$ ) denotes the membrane filtration area; and  $\Delta t$  (h)  
166 represents filtration duration.

167 Chemical oxygen demand (COD) of the feed and the permeate was measured in  
168 triplicate with a COD analyzer (DR2000, Hach). The COD rejection rate ( $R$  %) was

169 calculated with equation (3).

$$170 \quad R = (1 - C_p / C_f) \times 100\% \quad (3)$$

171 R is the COD removal rate;  $C_f$  and  $C_p$  are COD concentrations of the feed and the  
172 permeate, respectively.

### 173 **Characterization of surface properties of metal oxide**

174 The surface properties of five metal oxides were investigated in terms of surface  
175 hydroxyl groups, hydrophilicity, surface charge, and adhesion energy for oil droplets.  
176 The surface hydroxyl groups were examined with Fourier Transform Infrared  
177 Spectroscopy (FTIR) equipped with an Attenuated Total Reflection module (ATR)  
178 (Nicolet 6700-continuum FT-IR Microscope, Thermo Scientific) with a procedure  
179 described in **Text S4 (SI)**. Isoelectric points of metal oxides were measured with a  
180 Zeta potential analyzer (Nano ZS, Malvern) as detailed in **Text S5 (SI)**. The  
181 hydrophilicity of metal oxide was characterized with contact angle measurement  
182 (Attension, KSV). Since the contact angle is influenced by surface roughness of the  
183 solid sample,<sup>32, 33</sup> the metal oxides were deposited on smooth silicon wafers  
184 (roughness < 0.5 nm) separately with PLD to minimize the roughness effect.<sup>34, 35</sup> The  
185 roughness of the deposited wafers was measured with Atomic Force Microscopy  
186 (AFM) (Agilent 5500, Agilent) to be below 10 nm (**Table S2, SI**). Contact angle of  
187 each deposited metal-oxide layer on the wafer was measured 10 times using a  
188 water-droplet volume of  $5 \pm 0.2 \mu\text{L}$ . Mean value of the contact angles was calculated,  
189 and the standard deviation was below  $2^\circ$ . Adhesion energy of the metal oxide for oil  
190 droplets was investigated also with AFM (Agilent 5500, Agilent). Details of the AFM

191 probe preparation, experimental setup, and adhesion energy calculation were provided  
192 in **Text S6 (SI)**.

## 193 **Results and Discussion**

### 194 **Fouling tendency of deposited ceramic membranes**

195 Dead-end ultrafiltration with periodical back flush was applied to investigate  
196 irreversible fouling of metal oxide-deposited ceramic membranes in the treatment of  
197 the emulsion. The deposited membranes have nearly the same pure-water permeability,  
198 surface roughness and microstructures as the virgin membrane (**Table S3 and S4,**  
199 **Figure S5 and S6, SI**). **Figure S7 (SI)** shows the permeate-flux decline of these  
200 membranes as a function of filtration duration. The deposited membranes all suffered  
201 permeate-flux decline of different extent during filtration, which can be ascribed to  
202 membrane fouling and concentration/polarization on the cake layer.<sup>36</sup> Interestingly,  
203 two clear stages of the permeate-flux decline were observed before reaching a  
204 quasi-steady stage which has a relatively steady permeate flux (**Figure S7, SI and**  
205 **Figure 1**). The initial stage had a relatively slow decline of the permeate flux, being  
206 followed by a rapid permeate-flux decline stage. The development of the membrane  
207 fouling during the emulsion treatment was hypothesized in **Scheme 1**. At the  
208 beginning of the filtration, small oil droplets could directly penetrate and stuck inside  
209 membrane pores, which partially blocks the membrane pore leading to a slow  
210 decrease of the permeate flux (i.e., the first stage). Meanwhile, some oil droplets also  
211 deposit on the membrane surface. Then, the oil droplets would “pile-up” on  
212 membrane surface, which promotes the collision and coalescence of oil droplets as the

213 filtration further proceeds.<sup>37, 38</sup> The membrane pore entrance will consequently be  
214 covered by the coalesced oil droplets leading to significant reduction of the effective  
215 pore size and a quick decline of the permeate flux (i.e., the second stage). Eventually,  
216 a compact oil layer of coalesced droplets spreads out on membrane surface and fully  
217 covers on the pore entrance as a “second membrane”.<sup>55</sup> This uniform oil layer  
218 dominates permeate-flux and oil rejection, and leads to a steady permeate flux (i.e.,  
219 quasi-steady stage).<sup>39, 40</sup> Since the same O/W emulsion was used, the oil layer formed  
220 on each deposited membrane would be similar to each other, which would therefore  
221 lead to a similar oil rejection. COD rejection rates of the five deposited membranes  
222 (in **Table 1**) further confirmed our hypothesis. 93.2%, 96.1%, and 93.0% of COD  
223 rejection (COD = 324 mg O<sub>2</sub>/L) were obtained by negatively charged TiO<sub>2</sub><sup>-</sup>, CeO<sub>2</sub><sup>-</sup>, and  
224 MnO<sub>2</sub><sup>-</sup> membranes, respectively, while positively charged Fe<sub>2</sub>O<sub>3</sub><sup>-</sup> and CuO-  
225 membranes rejected 95.4% and 96.5% COD, respectively. The COD rejection rates of  
226 the five metal oxide-deposited membranes were close to each other. In general, solute  
227 transmission can be enhanced if the membrane is oppositely charged.<sup>41, 42</sup> The similar  
228 COD rejection of these deposited membranes here can be ascribed to a similar and  
229 compact oil layer formed on the membrane surface.<sup>39</sup>

230 **Figure 2** presents normalized initial permeate fluxes (NIF<sub>n</sub>, n represents the  
231 filtration cycles) of the deposited membranes in each cycle of O/W emulsion  
232 ultrafiltration. The NIF<sub>n</sub> of all the membranes declined as the number of  
233 filtration/backwash cycle increased, indicating that hydraulic irreversible membrane  
234 fouling increased gradually. In the six filtration cycles, the deposited membranes

235 suffered different extents of irreversible fouling. Fe<sub>2</sub>O<sub>3</sub>-membrane showed the best  
236 filtration performance and lowest irreversible fouling (NIF<sub>6</sub> = 60%), whereas  
237 MnO<sub>2</sub>-membrane suffered the most severe fouling (NIF<sub>6</sub> = 34%) of all the membranes.  
238 In comparison, TiO<sub>2</sub>-, CuO-, and CeO<sub>2</sub>- membranes had moderate irreversible fouling  
239 (i.e., NIF<sub>6</sub> = 52%, 47%, and 44%, respectively). The irreversible fouling tendency of  
240 the deposited membranes follows the order of Fe<sub>2</sub>O<sub>3</sub>-membrane < TiO<sub>2</sub>-membrane <  
241 CuO-membrane < CeO<sub>2</sub>-membrane < MnO<sub>2</sub>-membrane.

#### 242 **Surface properties of the deposition metal oxides**

243 Since the surface roughness and structures are nearly the same, the significant  
244 difference in irreversible fouling extent of the deposited membranes is considered to  
245 be caused by distinct surface properties of the deposited metal oxides. To elucidate the  
246 relationship between metal oxide surface properties and the irreversible fouling  
247 tendency, these metal oxides were fully characterized in terms of their surface  
248 hydroxyl groups in water, hydrophilicity, surface charge, and adhesion energy for oil  
249 droplets.

250 **Surface hydroxyl groups** Surface hydroxyl groups of metal oxides (MeO-H) play  
251 an important role in influencing their surface properties. Hydroxyl groups can interact  
252 with water molecules through chemisorption or hydrogen-bond interactions.<sup>43</sup> In  
253 addition, MeO-H formed Me-OH<sub>2</sub><sup>+</sup> or Me-O<sup>-</sup> species through protonation/  
254 deprotonation at pH below or above pH<sub>pzc</sub> of the metal oxide.<sup>44</sup> The surface hydroxyl  
255 groups of the metal oxides can be analyzed with ATR-FTIR.<sup>45-47</sup> The stretching band

256 of MeO-H can be separated from that of the O-H of bulk water by using D<sub>2</sub>O as  
257 solvent, where the surface hydroxyl group is deuterated.<sup>45</sup> The surface MeO-D has  
258 been reported to have characteristic IR stretching vibration in the range of 2200-2800  
259 cm<sup>-1</sup>.<sup>43, 46, 48</sup> **Figure 3** shows the ATR-FTIR spectra of the surface MeO-D groups of  
260 the metal oxides in D<sub>2</sub>O suspensions. The stretching frequencies of MeO-D of MnO<sub>2</sub>,  
261 CuO, CeO<sub>2</sub>, TiO<sub>2</sub>, and Fe<sub>2</sub>O<sub>3</sub> were 2337, 2339, 2456, 2545, and 2628 cm<sup>-1</sup>,  
262 respectively, indicating that the bond strength of O-D in MeO-D groups of the metal  
263 oxides follows the order of MnO<sub>2</sub> < CuO < CeO<sub>2</sub> < TiO<sub>2</sub> < Fe<sub>2</sub>O<sub>3</sub>. A strong O-D bond  
264 of the MeO-D group indicates that the metal site (Me) has relatively weak bonding  
265 with the O atom of the surface O-D group. The strong O-D bond also implies high  
266 electron density of the O-D group and thus possibly high affinity for D<sub>2</sub>O molecules  
267 through forming strong hydrogen bond. Therefore, the surface hydroxyl groups of  
268 Fe<sub>2</sub>O<sub>3</sub> probably tend to strongly interact with water molecules through hydrogen bond  
269 (i.e. highly hydrophilic), whereas those of MnO<sub>2</sub> and CuO would be significantly less  
270 hydrophilic. This hypothesis is confirmed in the following hydrophilicity test.

271 **Hydrophilicity** Hydrophilicity is an important parameter for predicting fouling  
272 tendency of polymeric membranes: high hydrophilicity usually means low membrane  
273 fouling potential in filtration.<sup>49, 50</sup> Metal oxides have been considered to be all  
274 hydrophilic. Differences in the hydrophilicity of metal oxides have never been  
275 realized until a recent study disclosed that some lanthanide oxides are relatively more  
276 hydrophobic than others.<sup>35</sup> The hydrophilicity of the metal oxides used in this study  
277 was tested with contact angle measurement. In general, a small contact angle indicates

278 a high hydrophilicity of the material.<sup>51</sup> Contact angle images of the five metal  
279 oxide-deposited wafers taken within 30 s of interaction with water droplets clearly  
280 show the differences of their hydrophilicity (**Figure 4a**). The contact angles of the  
281 five metal oxides varied significantly from 26.6° to 71.7° (**Table 1**). The  
282 hydrophilicity of the five metal oxides increases with the order of  $\text{MnO}_2 < \text{CeO}_2 <$   
283  $\text{CuO} < \text{TiO}_2 < \text{Fe}_2\text{O}_3$  (contact angles being 71.7°, 63.6°, 52.8°, 29.8°, 26.6°,  
284 respectively). This result supports the infer drawn with the order of IR stretching  
285 frequencies of the surface MeO–D bonds of these metal oxides, suggesting that the  
286 properties of surface hydroxyl groups largely influence the hydrophilicity of these  
287 metal oxides. **Figure 4b** schematically describes how water molecules attach on metal  
288 oxides by interaction with surface hydroxyl groups. A monolayer of water molecules  
289 interact directly with the surface hydroxyl groups of the metal oxide through  
290 hydrogen bonding, and then interact with outer-sphere water molecules also through  
291 hydrogen bonding forming multiple layers of water molecules.

292 **Surface charge** Surface charge of membrane material influences electrostatic  
293 interactions between the membrane and the solutes. Membrane fouling can be  
294 enhanced or reduced through electrostatic interactions.<sup>18, 30</sup> As described above, the  
295 surface charge of metal oxides is developed through protonation/deprotonation of  
296 surface hydroxyl groups, which is pH dependent.<sup>44</sup> At pH 6.0 of the experiments (i.e.,  
297 the pH of the emulsion),  $\text{MnO}_2$ ,  $\text{CeO}_2$ , and  $\text{TiO}_2$  ( $\text{pH}_{\text{pzc}}$  being 3.8, 4.2, and 3.2,  
298 respectively, as shown in **Table 1**) were negatively charged, while  $\text{Fe}_2\text{O}_3$  and  $\text{CuO}$   
299 ( $\text{pH}_{\text{pzc}}$  being 6.4 and 9.5, respectively) were positively charged. The surface charge of

300 metal oxide-deposited membranes was consistent with that of the deposition metal  
301 oxides at pH 6.0 (**Figure S8, SI**). Because oil droplets were negatively charged  
302 (**Table S1, SI**), electrostatic repulsion between oil droplets and particles of MnO<sub>2</sub>,  
303 CeO<sub>2</sub>, and TiO<sub>2</sub> would occur, while electrostatic attraction would exist between oil  
304 droplets and particles of Fe<sub>2</sub>O<sub>3</sub> and CuO.

305 **Adhesion energy for oil droplets** The absorption fouling of oil on metal oxides is  
306 not only influenced by hydrophobic-hydrophilic interactions, hydrogen bonding and  
307 electrostatic interactions, but also by van der Waals interactions between metal oxide  
308 and oil foulant.<sup>52</sup> AFM is a powerful tool that quantitatively measures the interaction  
309 forces/energies at the sub-nano Newton resolution between foulants and metal  
310 oxides.<sup>53, 54</sup> To further study the oil fouling potential towards metal oxides, adhesion  
311 energy of metal oxides for oil droplets was investigated with AFM. **Figure 5** shows  
312 that the least hydrophilic MnO<sub>2</sub> has the highest adhesion energy for oil droplets,  
313 whereas the most hydrophilic Fe<sub>2</sub>O<sub>3</sub> has the lowest adhesion energy of oil droplets.  
314 Adhesion energies of the metal oxides for oil droplets followed the order of Fe<sub>2</sub>O<sub>3</sub> <  
315 TiO<sub>2</sub> < CeO<sub>2</sub> < CuO < MnO<sub>2</sub>. It was noticed also that CuO had higher adhesion  
316 energy for oil droplets than CeO<sub>2</sub> while its hydrophilicity was higher. This means that  
317 the adhesion between oil droplets and metal oxides is also contributed by electrostatic  
318 interaction (CuO is oppositely charged to oil droplets, while CeO<sub>2</sub> is similarly charged  
319 to oil droplets).<sup>52</sup> Lower adhesion energy between oil droplets and metal oxide  
320 indicates lower oil adsorption tendency on the metal oxide. Accordingly, adsorption  
321 fouling of oil on metal oxides would decrease with the decreasing adhesion energy in

322 order of  $\text{Fe}_2\text{O}_3 < \text{TiO}_2 < \text{CeO}_2 < \text{CuO} < \text{MnO}_2$ , which can be inversely correlated to  
323 the hydrophilicity of the metal oxides.

324 **Relationship between the surface properties of filtration-layer metal oxide and**  
325 **the irreversible membrane fouling tendency**

326 The irreversible fouling tendency of the five metal oxide-deposited membranes can  
327 be correlated very well with surface properties of the metal oxides. The bond strength  
328 of surface hydroxyl groups of the metal oxides ( $\text{MeO-H}$ ) largely influences their  
329 hydrophilicity, and the hydrophilicity mainly dominates the adsorption fouling of oil  
330 on the metal oxides. The hydrophilicity of the metal oxides ( $\text{MnO}_2 < \text{CeO}_2 < \text{CuO} <$   
331  $\text{TiO}_2 < \text{Fe}_2\text{O}_3$ ) correlates inversely with irreversible fouling tendency of the deposited  
332 membranes, and higher hydrophilicity is related to lower irreversible fouling tendency.  
333 This result indicates that the hydrophilicity of the filtration-layer metal oxide is a  
334 major factor influencing irreversible fouling tendency of ceramic membrane in the  
335 O/W emulsion treatment. It is already known that hydrophilic polymeric membrane  
336 has low fouling tendency.<sup>55,56</sup> For ceramic membrane, this is the first time observing  
337 the difference of these common metal oxides in hydrophilicity and clearly revealing  
338 the dominance of the relative hydrophilicity of filtration-layer metal oxide in  
339 membrane fouling tendency.

340 Surface charge of the metal oxide opposite to that of the emulsion helps to alleviate  
341 irreversible membrane fouling in ultrafiltration, which is consistent with our last work  
342 regarding the influence of surfactant on ceramic membrane performance in O/W  
343 emulsion treatment.<sup>30</sup> The positively charged CuO-membrane suffered less

344 irreversible fouling than the negatively charged CeO<sub>2</sub>-membrane in the treatment of  
345 negatively charged emulsion (**Figure 2**), whereas its adsorption energy for oil was  
346 higher than that of CeO<sub>2</sub> (**Figure 5**). Irreversible membrane fouling is caused by 1)  
347 pore blockage by tiny oil droplets and 2) strong adsorption of oil droplets on  
348 membrane surface and pores.<sup>20</sup> Although the CuO-membrane has more adsorption  
349 fouling than the CeO<sub>2</sub>-membrane, it may have less pore blockage fouling and thus  
350 overall less irreversible fouling. Cross-section SEM images of the virgin and the  
351 fouled CuO- and CeO<sub>2</sub>- membranes support this hypothesis (**Figure S9, SI**). The  
352 oxide particles in the filtration layer of CuO- and CeO<sub>2</sub>- membranes (**Figure S9b and**  
353 **S9c, SI**) showed larger sizes than that of the virgin membrane (**Figure S9a, SI**). This  
354 indicates that oil droplets penetrated into membrane pores and covered on the oxide  
355 particles during filtration. The oxide particle size of the CuO-membrane was smaller  
356 than that of CeO<sub>2</sub>-membrane (**Figure S9b and S9c, SI**), indicating less oil droplet  
357 penetration into the CuO-membrane pores and covering on the oxide particles. Less  
358 pore blockage fouling of positively charged CuO-membrane could probably be  
359 ascribed to 1) the adsorption of some negatively charged surfactant-SDBS molecules  
360 on membrane surface and pores, which prevent the penetration of some oil droplets  
361 into the pores (i.e., surfactant barrier),<sup>42</sup> and 2) the subsequent coalescence of oil  
362 droplets near membrane surface because of the less stabilization SDBS for oil droplets  
363 (i.e., demulsification), which leads to more large oil droplets easily rejected on  
364 membrane surface and less small oil droplets available to penetrate into membrane  
365 pores (**Scheme S1, SI**).

366 It should be noticed that although the CuO-membrane oppositely charged to oil  
367 droplets may suffer less irreversible fouling caused by pore blockage, it still has  
368 overall more irreversible fouling than the TiO<sub>2</sub>-membrane similarly charged to oil  
369 droplets. This is probably because the CuO-membrane suffered much more  
370 irreversible adsorption fouling than the more hydrophilic TiO<sub>2</sub>-membrane (according  
371 to the adhesion energy shown in **Figure 5**). This fact again confirms that the  
372 hydrophilicity is a more significant factor influencing the irreversible fouling  
373 tendency of the ceramic membrane in O/W emulsion treatment.

374 **Environmental significance:**

375 Ceramic membrane ultrafiltration is an ideal treatment process for O/W emulsions  
376 daily produced in many industrial activities. Severe irreversible membrane fouling  
377 compromises the performance of ceramic membrane in O/W emulsion treatment,  
378 where filtration-layer material plays a crucial role. The filtration-layer material of low  
379 fouling tendency is highly desirable for sustainable O/W emulsion treatment. This  
380 study clearly shows the relationship between the surface properties of the  
381 filtration-layer metal oxides and the membrane fouling tendency, which provides a  
382 theoretical foundation to guide the selection/preparation/modification of ceramic  
383 membranes for sustainable O/W emulsion treatment. In addition, the result suggests  
384 that highly hydrophilic metal oxides (e.g., Fe<sub>2</sub>O<sub>3</sub>) can be potential filtration-layer  
385 materials of the ceramic membrane for O/W emulsion treatment.

386 **ASSOCIATED CONTENT**

387 **Supporting Information.**

388 Six texts, four tables, ten figures and one scheme are included in the supporting  
389 information. This information is available free of charge via the Internet at  
390 <http://pubs.acs.org/>

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544

545 Table 1. Surface properties of the metal oxides and COD rejection rates of ceramic  
 546 membranes deposited with these metal oxides for the O/W emulsion

Metal oxide	Contact angle (degrees)	pH <sub>pzc</sub>	Surface charge <sup>a</sup>	COD rejection %
Fe <sub>2</sub> O <sub>3</sub>	26.6 ± 1.6	6.4	+	95.4
CuO	52.8 ± 1.6	9.5	+	93.2
TiO <sub>2</sub>	29.8 ± 1.5	3.2	-	96.5
CeO <sub>2</sub>	63.6 ± 1.9	4.2	-	96.1
MnO <sub>2</sub>	71.7 ± 1.1	3.8	-	93.0

547 <sup>a</sup> Surface charge of metal oxides under experimental filtration conditions (pH = 6.0).

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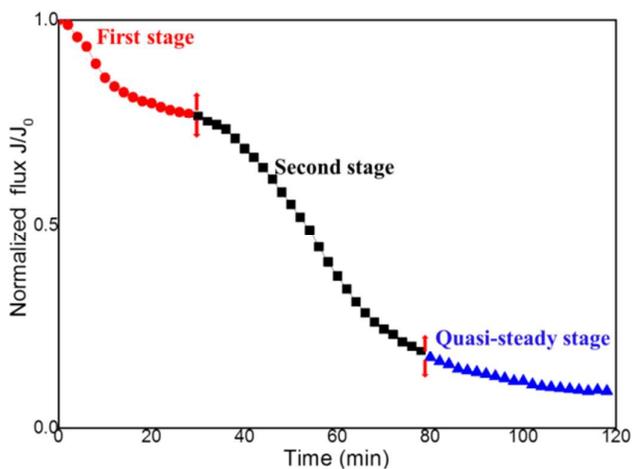
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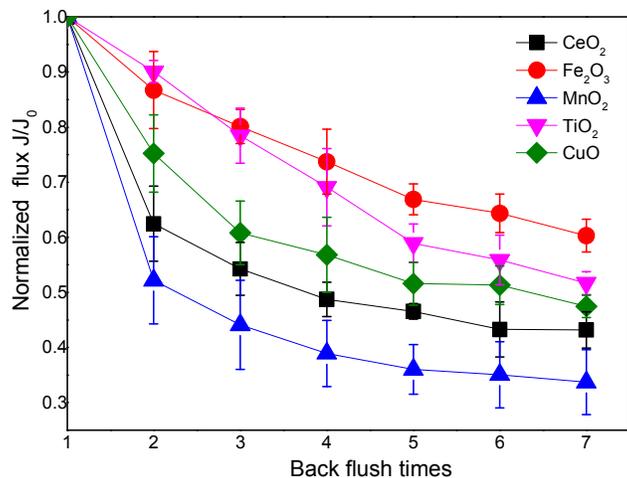


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556 Figure 1. Normalized permeate-flux decline of Fe<sub>2</sub>O<sub>3</sub>-membrane for O/W emulsion treatment in a long  
 557 run without back flush

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561 Figure 2. Decline of normalized initial permeate-flux of ceramic membranes deposited with different

562 metal oxides in seven cycles of O/W emulsion filtration.

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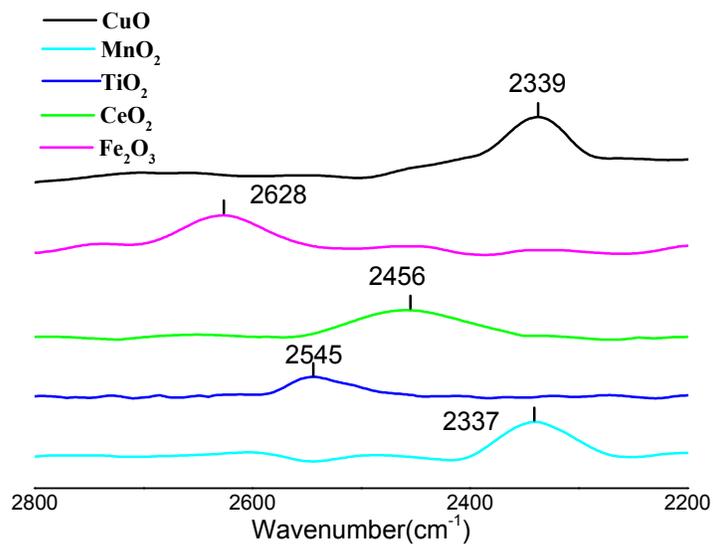
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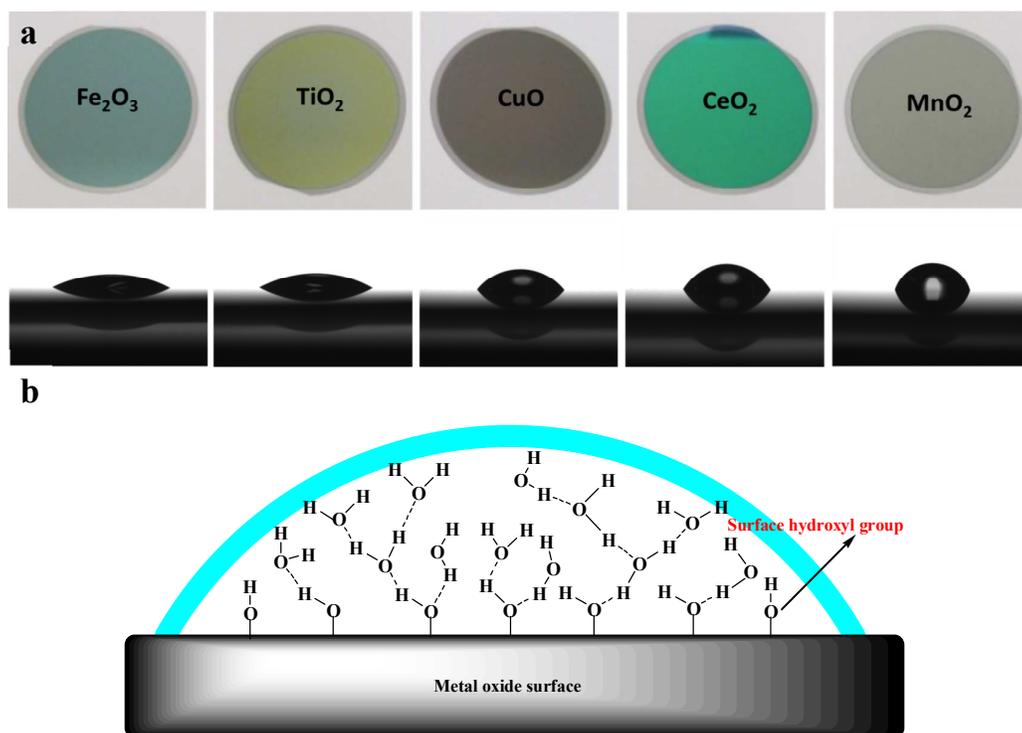
572 Figure 3. ATR-FTIR spectra of surface MeO-D groups of metal oxides immersed in D<sub>2</sub>O (pD = 6.0).

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Figure 4. Optical images and contact angle images of metal oxides deposited on silicon wafers (a). A

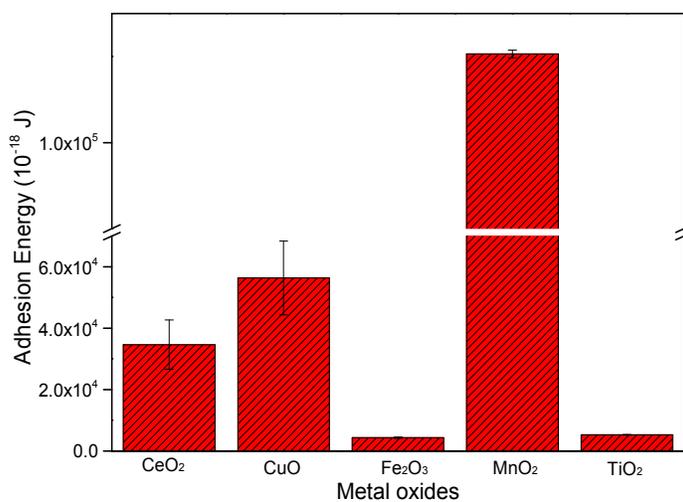
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scheme of interaction between water droplet and metal oxide surface (b).

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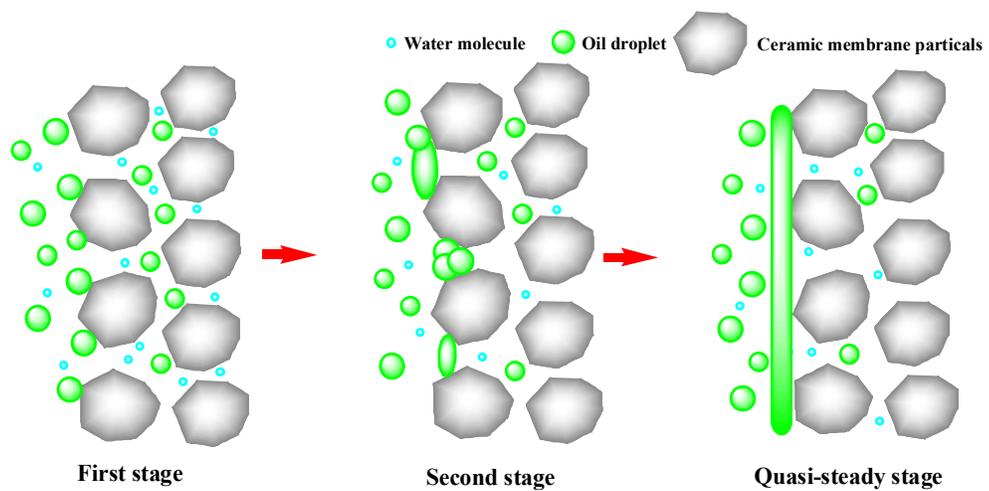
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Figure 5. Adhesion energy between metal oxides and oil droplets.

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591 Scheme 1. Possible membrane fouling stages of ceramic membrane during O/W emulsion treatment.

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