

# ACS NANO

Article



# ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

~~Two-Photon~~

Absorption in

# ACS Publications



ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

Organometallic

Bromide Perovskites



ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

Grant Walters, Brandon R  
Sutherland, Sjoerd Hoogland,  
Dong Shi, Riccardo Comin,



ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

Daniel P. Sellan, Osman M. Bakr, and Edward H. Sargent



ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

*ACS Nano*, **Just Accepted**  
Manuscript • DOI: 10.1021/



ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

acsnano.5b03308 • Publication

Date (Web): 21 Jul 2015



ACS Nano is published  
by the American  
Chemical Society. 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

# ACS Publications

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

Downloaded from <http://pubs.acs.org> on July 26, 2015



# ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright



# ACS NANO

Just Accepted



ACS Nano is published  
by the American  
Chemical Society. 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

# ACS Publications

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

# ACS NANO

"Just Accepted" manuscripts have been posted online prior to technical editing, formatting, and proofreading.



## ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

ACS NANO

Society provides "Just Accepted" as a  
dissemination of scientific material as soon as possible after the  
manuscript is accepted for publication. Just Accepted articles may  
appear in full in PDF format accompanied by abstracts and  
summaries.



ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

ACS NANO

fully peer reviewed, but should not be considered  
readers and citable by the Digital Object Identifier  
to authors. Therefore, the "Just Accepted"



ACS Publications

ACS Nano is published  
by the American  
Chemical Society, 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

Published by American  
Chemical Society.

Copyright © American  
Chemical Society.

However, no copyright

ACS NANO

in the journal. After a manuscript is technically accepted, it will appear on ACS Nano's "Accepted" Web site and published as an article in the journal. Any changes to the manuscript text and/or graphics must be made before the manuscript is accepted for publication.



ACS Nano is published by the American Chemical Society, 1155 Sixteenth Street N.W., Washington, DC 20036

Published by American Chemical Society.

Copyright © American Chemical Society.

However, no copyright

ACS NANO

and ethical guidelines that apply to the  
or consequences arising from the use of



ACS Nano is published  
by the American  
Chemical Society. 1155  
Sixteenth Street N.W.,  
Washington, DC 20036

ACS Publications

Published by American  
Chemical Society.

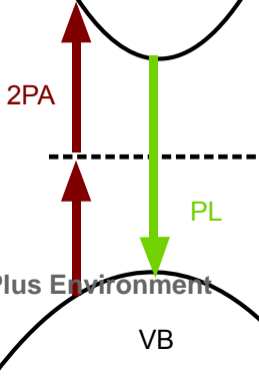
Copyright © American  
Chemical Society.

However, no copyright



ACS Nano

CB



ACS Paragon Plus Environment

# Two-Photon Absorption in Organometallic Bromide Perovskites

*Grant Walters,<sup>1†</sup> Brandon R. Sutherland,<sup>1†</sup> Sjoerd Hoogland,<sup>1</sup> Dong Shi,<sup>2</sup> Riccardo Comin,<sup>1</sup> Daniel P. Sellan,<sup>1</sup> Osman M. Bakr,<sup>2</sup> Edward. H. Sargent<sup>1\*</sup>*

<sup>1</sup> Department of Electrical and Computer Engineering, University of Toronto, Toronto, Ontario M5S 3G4, Canada

<sup>2</sup> Division of Physical Sciences and Engineering, Solar and Photovoltaics Engineering Research Center, King Abdullah University of Science and Technology (KAUST), Thuwal 23955-6900, Saudi Arabia

KEYWORDS: perovskite,  $\text{CH}_3\text{NH}_3\text{PbBr}_3$ , two-photon absorption, photoconductor, autocorrelator

ABSTRACT: Organometallic trihalide perovskites are solution processed semiconductors that have made great strides in third generation thin film light harvesting and light emitting optoelectronic devices. Recently it has been demonstrated that large, high purity single crystals of these perovskites can be synthesized from the solution phase. These crystals' large dimensions, clean bandgap, and solid-state order, have provided us with a suitable medium to observe and quantify two-photon absorption in perovskites. When  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  single crystals



1  
2  
3 are pumped with intense 800 nm light, we observe band-to-band photoluminescence at 572 nm,  
4  
5 indicative of two-photon absorption. We report the nonlinear absorption coefficient of  
6  
7  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  perovskites to be  $8.6 \text{ cm GW}^{-1}$  at 800 nm, comparable to epitaxial single crystal  
8  
9 semiconductors of similar bandgap. We have leveraged this nonlinear process to electrically  
10  
11 autocorrelate a 100 fs pulsed laser using a two-photon perovskite photodetector. This work  
12  
13 demonstrates the viability of organometallic trihalide perovskites as a convenient and low-cost  
14  
15 nonlinear absorber for applications in ultrafast photonics.  
16  
17  
18  
19

20  
21 Though formed at low temperature from the solution phase, organometallic halide perovskites  
22  
23 possess exceptional material properties. Perovskite thin films exhibit defect densities in the range  
24  
25  $10^{16} - 10^{17} \text{ cm}^{-3}$ ,<sup>1</sup> superior to other solution-cast semiconductors processed at similar  
26  
27 temperatures. Consistent with the low trap state density are the reported long and balanced  
28  
29 diffusion lengths,<sup>2</sup> large carrier mobilities,<sup>3,4</sup> and a linear absorption coefficient exceeding  $10^4$   
30  
31  $\text{cm}^{-1}$  at the band edge.<sup>5</sup> Driven by these remarkable material properties, perovskite thin film  
32  
33 absorbers have demonstrated record solution-processed solar cell power conversion efficiencies  
34  
35 up to a certified 20.1%.<sup>6</sup> Perovskite thin films have also been leveraged in light-emitting devices,  
36  
37 demonstrating low-threshold optically pumped lasers<sup>1,7,8</sup> and high-brightness light-emitting  
38  
39 diodes.<sup>9</sup>  
40  
41  
42  
43  
44

45  
46 Recently, there have been breakthrough demonstrations of the solution-processed growth of  
47  
48 large perovskite single crystals.<sup>10,11</sup> These single crystals offer millimeter-scale dimensions with  
49  
50 bulk trap state densities in the range  $10^9 - 10^{10} \text{ cm}^{-3}$ , comparable to some of the best epitaxial  
51  
52 single crystal semiconductors. The impressive energetic landscape in perovskite single crystals  
53  
54 has already been functionalized into ultra-low threshold optically pumped lasers.<sup>12,13</sup>  
55  
56  
57  
58  
59  
60

1  
2  
3 We sought to investigate the two-photon absorption in  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  perovskites and to  
4 demonstrate its potential in low-cost solution processed nonlinear optics. In addition to the  
5 importance of understanding the response of organometallic hybrid perovskites under intense  
6 radiation, their development as nonlinear optical materials and devices is a new opportunity.<sup>14</sup>  
7  
8 Semiconductor nonlinear absorbers have applications in optical limiting,<sup>15</sup> characterization of  
9 ultrafast optical signals,<sup>16</sup> lithography,<sup>17</sup> and microscopy.<sup>18</sup>  
10  
11  
12  
13  
14  
15  
16  
17

18 Equipped with the newly-discovered process to grow large single crystals, we were able to  
19 overcome many existing experimental challenges associated with quantifying two-photon  
20 absorption. We hypothesized that the large dimensions of single crystals, as well as their ultra-  
21 clean bandgaps and long range order would be an ideal material to study perovskite two-photon  
22 absorption. The small path length and dominant scattering of conventional perovskite thin films,  
23 which proved to limit the accuracy of these measurements, would no longer be an issue. Here we  
24 present an accurate quantification of two-photon absorption in  $\text{CH}_3\text{NH}_3\text{PbBr}_3$ . We demonstrate  
25 its applicability in nonlinear optics by autocorrelating a 100 fs laser using a perovskite two  
26 photon photodetector. This work is the first report of two-photon absorption and autocorrelation  
27 from a solution-processed single crystal organic-inorganic semiconductor.  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42

## 43 RESULTS AND DISCUSSION

44  
45

46 To study  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  perovskite, we employed the same vapor-assisted single crystal  
47 growth technique reported by Shi *et al.*<sup>10</sup> Here, the gradual diffusion of an antisolvent into a  
48 solution of perovskite precursors promotes seed formation and subsequent crystal growth. The  
49 resulting crystals are orange in color with regular rectangular facets (see Figure 1a inset). The  
50 large dimensions of the crystals we used enabled accurate measurements of two-photon  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

1  
2  
3 absorption, the magnitude of which is directly impacted by the propagation distance through the  
4  
5 sample.  
6  
7

8  
9 Two-photon absorption is a third-order nonlinear process that becomes relevant at high photon  
10  
11 fluences. Two photons, each with energy less than the difference between the material's ground  
12  
13 and excited states, are simultaneously absorbed and excite a charge carrier. In semiconductors,  
14  
15 this is manifested as an excitation across the bandgap. Two-photon absorption varies with the  
16  
17 square of the electric field intensity  $I$ . The attenuation of light propagating a distance  $z$  through a  
18  
19 two-photon absorbing medium can thus be written as,  
20  
21

$$\frac{dI}{dz} = -\alpha I - \beta I^2, \quad (1)$$

22  
23  
24 where  $\alpha$  and  $\beta$  are the one- (linear) and two-photon absorption coefficients respectively. The  
25  
26 solution of this differential equation, expressing the transmission  $T$  through a sample of length  $L$ ,  
27  
28 is:  
29  
30  
31  
32  
33

$$T = \frac{I}{I_0} = \frac{e^{-\alpha L}}{\beta I_0 L_{eff} + 1}, \quad (2)$$

34  
35  
36 where  $I_0$  is the peak intensity of light entering the sample and  $L_{eff}$  is the effective sample length  
37  
38 given by,  
39  
40  
41  
42  
43  
44

$$L_{eff} = \frac{1 - e^{-\alpha L}}{\alpha}. \quad (3)$$

45  
46  
47 The intensity of light entering the sample is found by reducing the measured incident intensity by  
48  
49 a factor of  $(1 - R)$  where  $R$  is the surface reflectivity.  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

1  
2  
3 In order to achieve degenerate two-photon absorption in our samples we used a high-power  
4 mode-locked titanium sapphire laser operating at 800 nm to provide below-bandgap excitation.  
5  
6 The bandgap of  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  is 2.21 eV,<sup>10</sup> with an absorption onset at 560 nm as shown in  
7  
8 Figure 1a. This onset is sharp, leading to nearly constant absorption for above-bandgap radiation.  
9  
10 The step-like behavior in the absorption ensured the samples were free from linear absorption of  
11  
12 the incident laser light. Previous transient absorption measurements on  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  single  
13  
14 crystals using above-bandgap pump and below-bandgap probe lasers have shown significant  
15  
16 changes in optical absorption within picoseconds of excitation; however, the excited population  
17  
18 that allows for this change in absorption quickly decays such that the material regains its below-  
19  
20 bandgap transparency within nanoseconds.<sup>10</sup> We have used ultra-short pulses (100 fs) with a  
21  
22 periodicity of 13 ns in our experiments. The use of ultra-short pulses in our experiments and the  
23  
24 absence of linear absorption means photoinduced absorption associated with excited-state  
25  
26 carriers was negligible and did not affect the accuracy of our measurements.  
27  
28  
29  
30  
31  
32  
33  
34

35 Strikingly bright green light is visibly emitted when the samples are placed in the path of the  
36  
37 laser (Figure 1a inset). This photoluminescence was direct evidence that our samples were in-fact  
38  
39 absorbing the below-bandgap 800 nm radiation through two-photon absorption. The resulting  
40  
41 emission spectrum in Figure 1a shows a sharp and narrow peak at 572 nm—with a wavelength  
42  
43 dependence unchanged from linearly induced photoluminescence. Thus the perovskite crystals  
44  
45 behave as an effective upconverter for near infrared radiation. A simple schematic of the two-  
46  
47 photon absorption and induced photoluminescence process is given in Figure 1b.  
48  
49  
50  
51

52 The z-scan procedure<sup>19</sup> is the most widely-used method for measuring two-photon absorption  
53  
54 coefficients because of its simplicity and ease of data interpretation. It involves translating the  
55  
56 sample along the path of a focused laser beam in order to vary the incident intensity and find  $\beta$ .  
57  
58  
59  
60

1  
2  
3 Although convenient, the method can be prone to artefacts and overestimation with less-than-  
4 ideal samples. To avoid possible changes in scattering associated with changes in the spot-size as  
5 the sample is translated for z-scans, we opted to alternatively measure the two-photon absorption  
6 coefficient for  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  through static intensity-dependent transmission measurements.  
7  
8 This straight-forward technique provides an accurate description of two-photon absorption  
9 behavior.<sup>20</sup> These measurements were made by placing the sample at the waist of the focused  
10 laser beam, varying the laser intensity with a filter wheel, and monitoring incident and  
11 transmitted power.  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22

23 A typical plot of inverse transmission *versus* peak intensity is given in Figure 2a. The plot  
24 shows the requisite linear increase with intensity according to equation 2 accompanied by an  
25 expected slight downward curvature, the latter behavior indicating temporal and spatial  
26 broadening of the beam towards the rear of the sample when subjected to high intensities.<sup>20</sup> The  
27 solid line is a least-squares fit according to the equations for temporally and spatially Gaussian  
28 pulses developed by Sheik-Bahae *et al.*<sup>19</sup>:  
29  
30  
31  
32  
33  
34  
35  
36  
37

$$T(I_0) = \sum_{m=0}^{\infty} \frac{[-q(I_0)]^m}{(m+1)^2}, \quad (4)$$

$$q(I_0) = \beta I_0 L_{eff}, \quad (5)$$

38  
39  
40  
41  
42  
43  
44  
45  
46  
47 The sum in equation 4 was performed for 20 terms to ensure proper convergence of the function.  
48  
49 Although linear absorption can be ignored for these samples, they often showed significant  
50 scattering and so this loss mechanism was treated as linear absorption in accordance with Ref.<sup>20</sup>.  
51  
52 Repeated measurements on several samples yielded an average absorption coefficient of  $8.6 \pm$   
53  
54  
55  
56  
57  
58  
59  
60  
61  
62  
63  
64  
65  
66  
67  
68  
69  
70  
71  
72  
73  
74  
75  
76  
77  
78  
79  
80  
81  
82  
83  
84  
85  
86  
87  
88  
89  
90  
91  
92  
93  
94  
95  
96  
97  
98  
99  
100  
101  
102  
103  
104  
105  
106  
107  
108  
109  
110  
111  
112  
113  
114  
115  
116  
117  
118  
119  
120  
121  
122  
123  
124  
125  
126  
127  
128  
129  
130  
131  
132  
133  
134  
135  
136  
137  
138  
139  
140  
141  
142  
143  
144  
145  
146  
147  
148  
149  
150  
151  
152  
153  
154  
155  
156  
157  
158  
159  
160  
161  
162  
163  
164  
165  
166  
167  
168  
169  
170  
171  
172  
173  
174  
175  
176  
177  
178  
179  
180  
181  
182  
183  
184  
185  
186  
187  
188  
189  
190  
191  
192  
193  
194  
195  
196  
197  
198  
199  
200  
201  
202  
203  
204  
205  
206  
207  
208  
209  
210  
211  
212  
213  
214  
215  
216  
217  
218  
219  
220  
221  
222  
223  
224  
225  
226  
227  
228  
229  
230  
231  
232  
233  
234  
235  
236  
237  
238  
239  
240  
241  
242  
243  
244  
245  
246  
247  
248  
249  
250  
251  
252  
253  
254  
255  
256  
257  
258  
259  
260  
261  
262  
263  
264  
265  
266  
267  
268  
269  
270  
271  
272  
273  
274  
275  
276  
277  
278  
279  
280  
281  
282  
283  
284  
285  
286  
287  
288  
289  
290  
291  
292  
293  
294  
295  
296  
297  
298  
299  
300  
301  
302  
303  
304  
305  
306  
307  
308  
309  
310  
311  
312  
313  
314  
315  
316  
317  
318  
319  
320  
321  
322  
323  
324  
325  
326  
327  
328  
329  
330  
331  
332  
333  
334  
335  
336  
337  
338  
339  
340  
341  
342  
343  
344  
345  
346  
347  
348  
349  
350  
351  
352  
353  
354  
355  
356  
357  
358  
359  
360  
361  
362  
363  
364  
365  
366  
367  
368  
369  
370  
371  
372  
373  
374  
375  
376  
377  
378  
379  
380  
381  
382  
383  
384  
385  
386  
387  
388  
389  
390  
391  
392  
393  
394  
395  
396  
397  
398  
399  
400  
401  
402  
403  
404  
405  
406  
407  
408  
409  
410  
411  
412  
413  
414  
415  
416  
417  
418  
419  
420  
421  
422  
423  
424  
425  
426  
427  
428  
429  
430  
431  
432  
433  
434  
435  
436  
437  
438  
439  
440  
441  
442  
443  
444  
445  
446  
447  
448  
449  
450  
451  
452  
453  
454  
455  
456  
457  
458  
459  
460  
461  
462  
463  
464  
465  
466  
467  
468  
469  
470  
471  
472  
473  
474  
475  
476  
477  
478  
479  
480  
481  
482  
483  
484  
485  
486  
487  
488  
489  
490  
491  
492  
493  
494  
495  
496  
497  
498  
499  
500  
501  
502  
503  
504  
505  
506  
507  
508  
509  
510  
511  
512  
513  
514  
515  
516  
517  
518  
519  
520  
521  
522  
523  
524  
525  
526  
527  
528  
529  
530  
531  
532  
533  
534  
535  
536  
537  
538  
539  
540  
541  
542  
543  
544  
545  
546  
547  
548  
549  
550  
551  
552  
553  
554  
555  
556  
557  
558  
559  
560  
561  
562  
563  
564  
565  
566  
567  
568  
569  
570  
571  
572  
573  
574  
575  
576  
577  
578  
579  
580  
581  
582  
583  
584  
585  
586  
587  
588  
589  
590  
591  
592  
593  
594  
595  
596  
597  
598  
599  
600  
601  
602  
603  
604  
605  
606  
607  
608  
609  
610  
611  
612  
613  
614  
615  
616  
617  
618  
619  
620  
621  
622  
623  
624  
625  
626  
627  
628  
629  
630  
631  
632  
633  
634  
635  
636  
637  
638  
639  
640  
641  
642  
643  
644  
645  
646  
647  
648  
649  
650  
651  
652  
653  
654  
655  
656  
657  
658  
659  
660  
661  
662  
663  
664  
665  
666  
667  
668  
669  
670  
671  
672  
673  
674  
675  
676  
677  
678  
679  
680  
681  
682  
683  
684  
685  
686  
687  
688  
689  
690  
691  
692  
693  
694  
695  
696  
697  
698  
699  
700  
701  
702  
703  
704  
705  
706  
707  
708  
709  
710  
711  
712  
713  
714  
715  
716  
717  
718  
719  
720  
721  
722  
723  
724  
725  
726  
727  
728  
729  
730  
731  
732  
733  
734  
735  
736  
737  
738  
739  
740  
741  
742  
743  
744  
745  
746  
747  
748  
749  
750  
751  
752  
753  
754  
755  
756  
757  
758  
759  
760  
761  
762  
763  
764  
765  
766  
767  
768  
769  
770  
771  
772  
773  
774  
775  
776  
777  
778  
779  
780  
781  
782  
783  
784  
785  
786  
787  
788  
789  
790  
791  
792  
793  
794  
795  
796  
797  
798  
799  
800  
801  
802  
803  
804  
805  
806  
807  
808  
809  
810  
811  
812  
813  
814  
815  
816  
817  
818  
819  
820  
821  
822  
823  
824  
825  
826  
827  
828  
829  
830  
831  
832  
833  
834  
835  
836  
837  
838  
839  
840  
841  
842  
843  
844  
845  
846  
847  
848  
849  
850  
851  
852  
853  
854  
855  
856  
857  
858  
859  
860  
861  
862  
863  
864  
865  
866  
867  
868  
869  
870  
871  
872  
873  
874  
875  
876  
877  
878  
879  
880  
881  
882  
883  
884  
885  
886  
887  
888  
889  
890  
891  
892  
893  
894  
895  
896  
897  
898  
899  
900  
901  
902  
903  
904  
905  
906  
907  
908  
909  
910  
911  
912  
913  
914  
915  
916  
917  
918  
919  
920  
921  
922  
923  
924  
925  
926  
927  
928  
929  
930  
931  
932  
933  
934  
935  
936  
937  
938  
939  
940  
941  
942  
943  
944  
945  
946  
947  
948  
949  
950  
951  
952  
953  
954  
955  
956  
957  
958  
959  
960  
961  
962  
963  
964  
965  
966  
967  
968  
969  
970  
971  
972  
973  
974  
975  
976  
977  
978  
979  
980  
981  
982  
983  
984  
985  
986  
987  
988  
989  
990  
991  
992  
993  
994  
995  
996  
997  
998  
999  
1000

1  
2  
3 CdSe ( $E_g = 1.74$  eV) have absorption coefficients of  $5.5 \text{ cm GW}^{-1}$  (at 530 nm) and  $18 \text{ cm GW}^{-1}$   
4 (at 1060 nm) respectively.<sup>21</sup> The dependence of two-photon absorption with band-gap for  
5  
6  
7  
8  
9  
10  
11  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

CdSe ( $E_g = 1.74$  eV) have absorption coefficients of  $5.5 \text{ cm GW}^{-1}$  (at 530 nm) and  $18 \text{ cm GW}^{-1}$  (at 1060 nm) respectively.<sup>21</sup> The dependence of two-photon absorption with band-gap for semiconductors is well established and we find our value for  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  agrees well with the reported scaling laws.<sup>20–22</sup> The experimental value has only a 14% error with our own prediction based on these relations:  $10 \text{ cm GW}^{-1}$ . Consideration of the corresponding scaling laws for the nonlinear refractive index and the Kramers-Kronig relation between the two-photon absorption coefficient and the nonlinear refractive index shows that the nonlinear reflection due to changes in the material's refractive index (a maximum change of  $10^{-5}$ ) is negligible and does not impact the measured two-photon absorption coefficient.<sup>22</sup>

The use of single crystals enabled us to measure the polarization dependence of two-photon absorption in the perovskites. These measurements are fundamentally impossible with polycrystalline thin films due to their random lattice orientations. Figure 2b shows the polarization dependence of two-photon absorption for  $\text{CH}_3\text{NH}_3\text{PbBr}_3$ . Plotted is the transmission scaled by the incident intensity  $T/I_{in}$  – a quantity directly proportional to the absorption coefficient. A half-wave plate was used to rotate the incident beam's electric field polarization an angle  $\theta$  relative to the [100] crystallographic axis and the incident propagation along the [001] direction such that the polarization rotated through the crystallographic xy plane.  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  is cubic with the space group  $P m\bar{3}m$  at room temperature,<sup>10</sup> and so the polarization dependence can be modeled with:

$$\frac{T}{I_{in}} = A[1 + 2\sigma[\sin^4(\theta + \varphi) - \sin^2(\theta + \varphi)]], \quad (6)$$

1  
2  
3 where  $A$  is related to the imaginary component of the susceptibility tensor element  $\chi_{xxxx}^{(3)}$ ,  $\sigma$  is  
4 the sample anisotropy parameter, and  $\varphi$  is the wave-plate angular offset.<sup>23–25</sup> A least-squares fit  
5  
6 is shown in Figure 2b and yields an anisotropy parameter of  $\sigma = -0.07$ .  
7  
8  
9

10  
11  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  and other organometallic perovskites have previously been predicted, through  
12 density functional theory calculations, to exhibit a large spin-orbit coupling due to the heavy lead  
13 atoms.<sup>26–31</sup> This coupling splits the degenerate conduction band, such that a doublet band is  
14 lowered and a quartet band is raised in energy.<sup>26</sup> These predictions are supported by an  
15 agreement between the experimentally measured and the predicted values for the bandgap. For  
16 semiconductors in general, the valence band to lowest conduction band transitions do not provide  
17 any orientational dependence for two-photon absorption.<sup>24,25,32</sup> Anisotropic behavior may be  
18 introduced either from direct transitions between the valence bands and the higher conduction  
19 bands or from state mixing between these bands.<sup>24,25,32</sup> These mechanisms are minimal in  
20  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  where the conduction bands have a large energetic separation, consistent with the  
21 small anisotropy parameter observed.  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38

39 We utilized the nonlinear properties of our perovskite single crystals for autocorrelation  
40 purposes. This can be done with measurements of the nonlinear response of transmission,  
41 photoluminescence intensity, second harmonic generation, or electrical current. In line with the  
42 facile processing of the perovskite crystals and their exceptional electrical properties we aimed to  
43 use electrical current, one of the most straightforward signals to measure, as our autocorrelation  
44 signal.  
45  
46  
47  
48  
49  
50  
51  
52  
53

54 We fabricated a simple photoconductor by depositing silver contacts on opposite ends of a  
55 single crystal. A schematic of the photodetector, along with the crystal dimensions is shown in  
56  
57  
58  
59  
60

1  
2  
3 Figure 3a. The sample is illuminated from the side, resulting in single-pass two-photon  
4 absorption of 800 nm light. The dark current-voltage characteristic of this crystal is shown in  
5  
6 Figure 3b. This crystal has an electrical conductivity of  $10^{-5} \Omega^{-1} \text{ cm}^{-1}$ .  
7  
8  
9

10  
11 The photocurrent (light minus dark current) generated as a result of single-pass absorption of  
12 800 nm light is given as a function of peak intensity of the pump laser at a fixed bias of 50 V in  
13 Figure 3c. We tested control samples of just glass and just silver and no photocurrent is  
14 observed. Photoconductors based on linear absorption have a photocurrent which increases  
15 sublinearly (power-law dependence  $n < 1$ ) with input intensity (see Supporting Information  
16 Figure S1a).<sup>33</sup> However, the two-photon absorption probability increases with the square of input  
17 intensity ( $n = 2$ ). We therefore expect a photocurrent dependence on input intensity which is less  
18 than  $n = 2$ , but greater than linear photoconduction ( $n < 1$ ), which is what we observe.  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30

31 The responsivity (efficiency in  $\text{A W}^{-1}$ ) of the two photon photodetector at a fixed bias of 50 V  
32 as a function of input intensity is given in Figure 3d. The responsivity is of order  $10^{-7}$ . This value  
33 is five orders of magnitude less than the upper limit in the absence of photoconductive gain as  
34 determined by the two-photon absorption coefficient. Low responsivity is expected given that  
35 perovskites, with their low trap state density and balanced carrier mobilities, exhibit low intrinsic  
36 photoconductive gain.<sup>34</sup> Despite the low gain, due to the large number of total absorbed photons,  
37 the device output photocurrent is in the range 10 nA – 1  $\mu\text{A}$ , readily measurable using a  
38 conventional current meter. The responsivity increases 3.5x over 1 decade of input intensity.  
39  
40 This is contrary to linear photoconductors, which have a responsivity that decreases with  
41 increasing input intensity (see Supporting Information Figure S1b). This upward trend  
42 demonstrates that the photocurrent is a result of nonlinear absorption and not linear processes  
43 such as thermal generation or trap-mediated absorption.  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60



1  
2  
3 A schematic of our perovskite autocorrelator is shown in Figure 4a. A 100 fs, 800 nm pulse  
4 passes through a 50/50 beam splitter (BS). One arm is sent to a retro-reflector on a delay line.  
5  
6 The two arms meet again at a second 50/50 splitter which sends half of the total light to a power  
7  
8 meter (det), the other half through a focusing lens onto the two-photon absorbing perovskite  
9  
10 photodetector. The generated photocurrent is collected using a current meter and monitored as a  
11  
12 function of the delay line position. The resulting current-time trace is shown in Figure 4b. Our  
13  
14 autocorrelator configuration is a modified Mach–Zehnder interferometer where the two pulses  
15  
16 have the same phase and polarization. With this configuration, we observe constructive and  
17  
18 destructive interference fringes corresponding to the overlap of the time-varying electrical fields.  
19  
20 The envelope of these fringes corresponds to the intensity overlap of the pulses in time. We fit  
21  
22 the top and bottom envelopes to a  $\text{sech}^2(t)$  function, shown in red. The resulting autocorrelation  
23  
24 has a full-width at half-maximum (FWHM) of 160 fs. Since we are autocorrelating the electric  
25  
26 field, but we wish to know the intensity profile, there is a deconvolution factor to be applied. For  
27  
28 a  $\text{sech}^2(t)$  input pulse intensity profile, the deconvolution factor is 0.65,<sup>35</sup> resulting in a measured  
29  
30 pulse duration of 104 fs. This is in excellent agreement with the 100 fs specification of our laser.  
31  
32 The fringe spacing, limited by the resolution of our motorized stage, is approximately every 3/2  
33  
34 of a wavelength, shown in Figure 4c.  
35  
36  
37  
38  
39  
40  
41  
42  
43

## 44 CONCLUSION

45  
46  
47 This work demonstrates the prospects of perovskites as solution processed two-photon  
48  
49 absorbers for applications in nonlinear optics. With just readily available chemical precursors,  
50  
51 researchers can fabricate large single crystal two-photon absorbers to accurately and  
52  
53 conveniently autocorrelate ultrafast lasers. Yet, the nonlinear optical properties of perovskites  
54  
55 remain largely unexplored. Future work will focus on investigating new avenues for perovskites  
56  
57  
58  
59  
60

1  
2  
3 as nonlinear optical materials, such as studying secondary harmonic generation, optical limiting,  
4  
5 and self-focusing.  
6  
7

## 8 9 METHODS

10  
11  
12  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  single crystals were prepared using a vapor-assisted crystallization procedure as  
13 previously reported.<sup>10</sup> In brief, the vapor from an antisolvent (dichloromethane, in our case) was  
14 allowed to slowly diffuse into a solution of perovskite precursors, lead bromide ( $\text{PbBr}_2$ ) and  
15 methylammonium bromide ( $\text{CH}_3\text{NH}_3\text{Br}$ ) in *N,N*-dimethylformamide, such that precursors  
16 spontaneously nucleated and co-crystallized in a self-sustained process. Crystals were kept in the  
17 mother liquor until use in optical experiments. Their dimensions were measured using a  
18 Mitutoyo digital caliper to within 10  $\mu\text{m}$ .  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29

30  
31 Linear optical absorption was characterized using a PerkinElmer Lambda 950 UV/Vis/NIR  
32 spectrophotometer in the wavelength range 250 – 900 nm and using 2 nm incremental steps.  
33  
34 Photoluminescence induced by linear absorption was measured using a Liconix Helium  
35 Cadmium 442 nm excitation laser and Ocean Optics USB2000 spectrometer.  
36  
37  
38  
39

40  
41 For all nonlinear absorption experiments, a mode-locked Mira 900-F Ti:Sapphire laser was  
42 used as the excitation source. The pulse duration was 100 fs (FWHM) at a repetition rate of 76  
43 MHz. The laser was tuned to an output wavelength of 800 nm and produced horizontally  
44 polarized  $\text{TEM}_{00}$ -mode single pulses. Power measurements were made with an Ophir LaserStar  
45 Dual Channel power and energy meter with photodiode sensors which had an accuracy of  $\pm 5\%$   
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60 after attenuation at 800 nm.

1  
2  
3 Two-photon absorption coefficients were measured by placing the sample at the focus of a lens  
4 with a 15 cm focal length and by attenuating the incident laser radiation with a neutral density  
5 filter wheel. The incident and transmitted light was measured with power meters placed before  
6 and after the sample. The lens focused the beam to a spot size of 35  $\mu\text{m}$  (half-width at  $1/e^2$ ) as  
7 determined with a CCD camera setup. For most measurements, the incident beam was chopped  
8 at 400 Hz in order to prevent sample heating and optical damage. To measure the polarization  
9 dependence of two-photon absorption, a rotating half-wave plate was placed in the path of the  
10 incident beam. Predicted absorption coefficients were determined through the scaling law,  
11  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23

$$\beta = 3100 \frac{\sqrt{E_P} (2\hbar\omega/E_g - 1)^{3/2}}{n^2 E_g^3 (2\hbar\omega/E_g)^5}, \quad (7)$$

24  
25  
26  
27  
28 where,

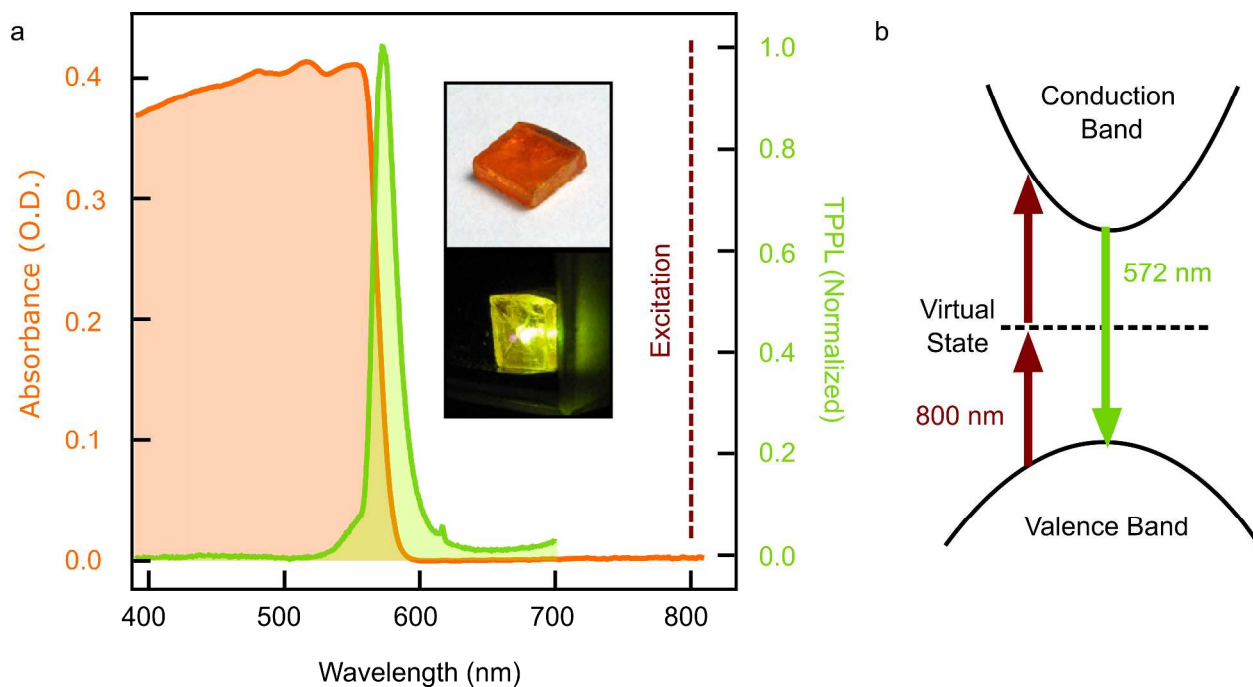
$$E_P = \frac{2P^2 m}{\hbar^2} \quad (\text{in eV}), \quad (8)$$

29  
30  
31  
32 and  $E_g$  is the bandgap (eV),  $\hbar\omega$  is the field energy (eV),  $m$  is the electron mass,  $n$  is the linear  
33 refractive index (assumed 2.5) and  $P$  is the Kane momentum parameter (assumed  $4.5 \times 10^{-27}$  J  
34 m).<sup>20-22</sup>  
35  
36  
37  
38  
39  
40  
41  
42  
43

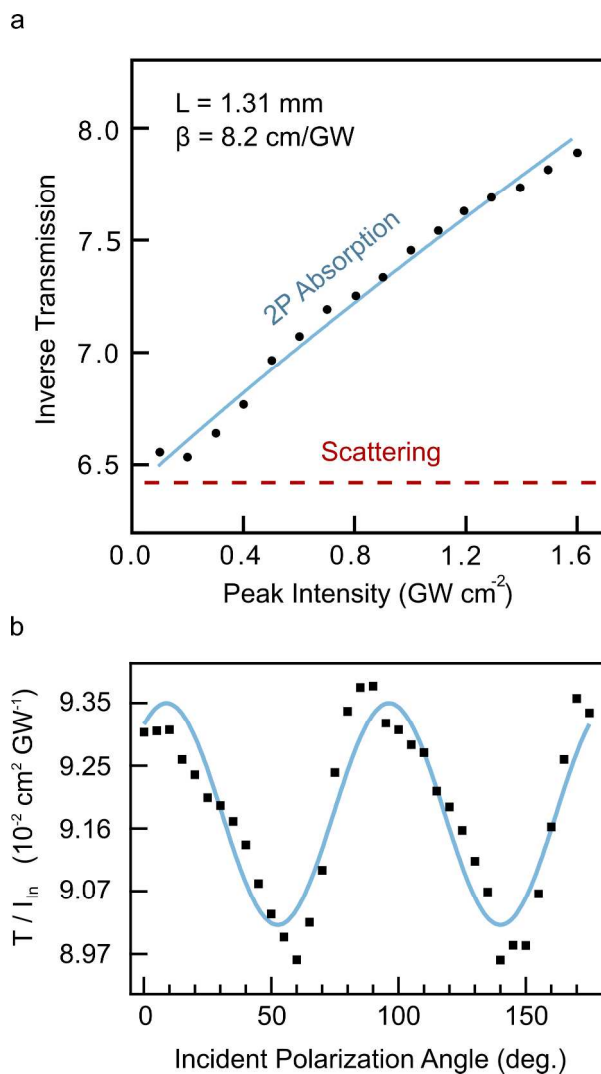
44 Photoconduction measurements were done using a device made from a single crystal with  
45 wires bonded to opposing crystal faces with conductive silver paste. Two-photon absorption  
46 photocurrent was measured using a Keithley 2410 High Voltage Source Meter (also used as a  
47 bias source).  
48  
49  
50  
51  
52  
53

54 For autocorrelation experiments the incident laser beam was split into two branches, such that  
55 a motorized translation stage with a retroreflector could be added to introduce a pulse-delay for  
56  
57  
58  
59  
60

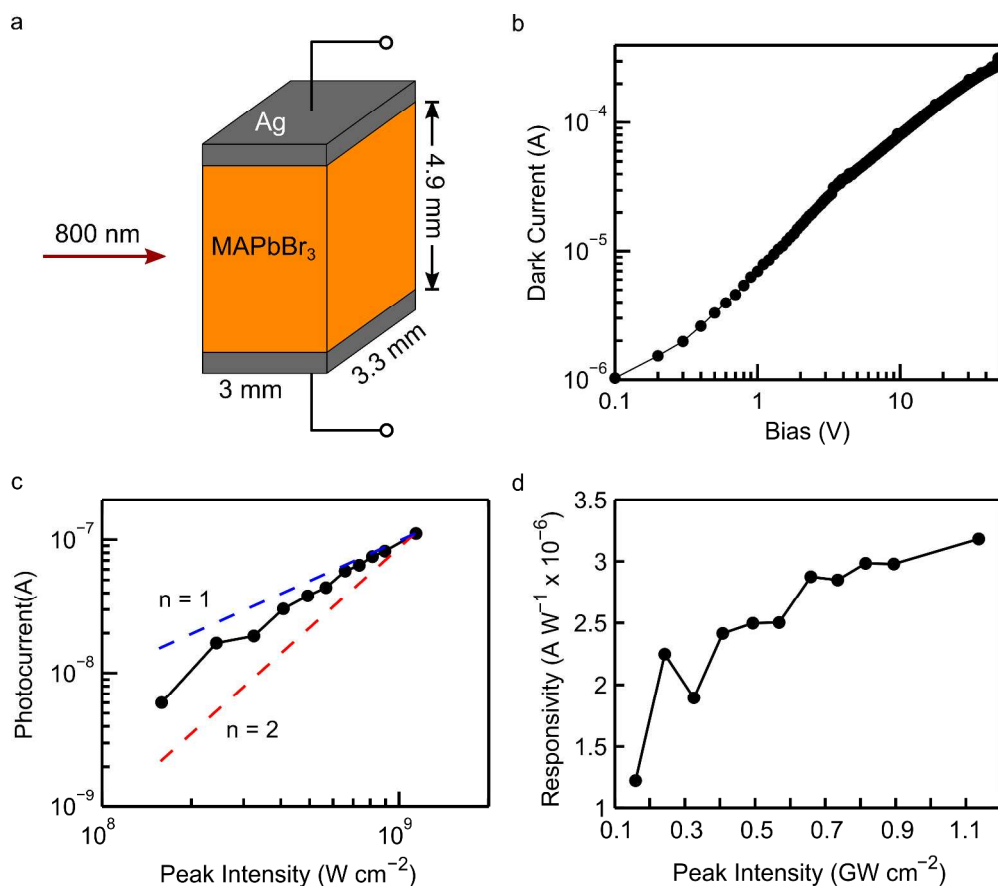
one of the paths. The separate paths were then rerouted into a single collinear path that was incident on our sample. Photocurrent was measured as a function of the retroreflector position in order to detect pulse-overlap and determine the autocorrelation width.



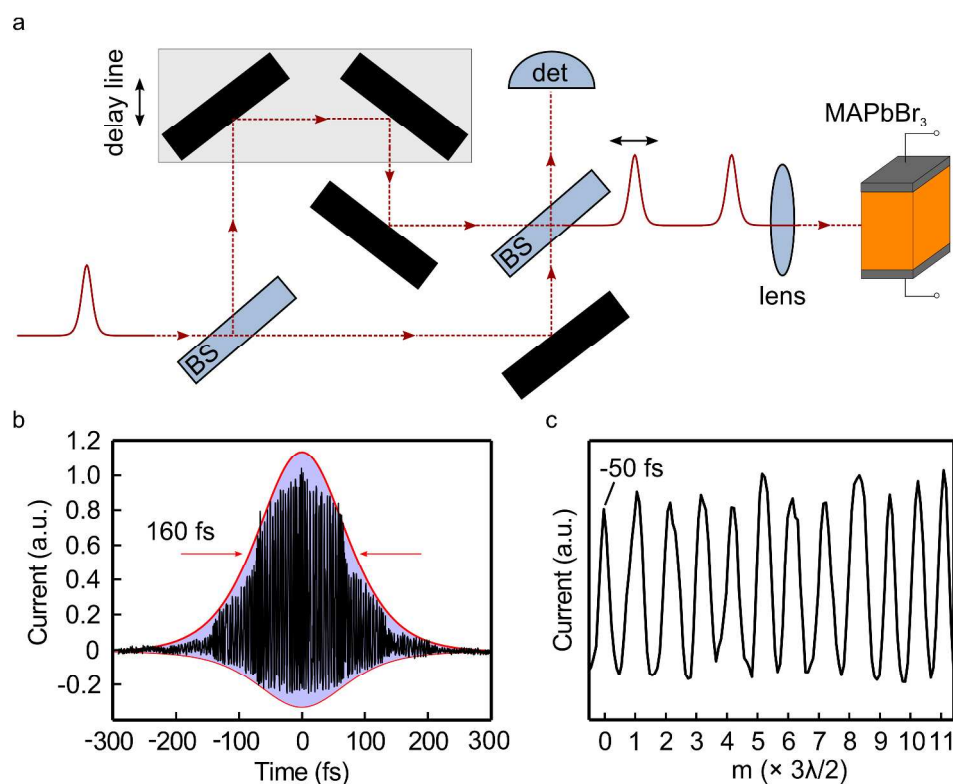
**Figure 1.** Optical properties. a) Linear optical absorbance (left axis) and normalized two-photon-induced photoluminescence (TPPL, right axis) as a function of wavelength for a single  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  crystal. Photoluminescence peaks at 572 nm. Insets show the crystal (dimensions  $\sim 1 \times 5 \times 5$  mm) under room light (upper) and brightly luminescing when exposed to intense 800 nm laser radiation (lower). The visibly observed photoluminescence is a direct signature of two-photon absorption. b) Schematic showing two-photon absorption of 800 nm light and up-conversion to 572 nm photoluminescence.



**Figure 2.** Two-photon absorption coefficient. a) Inverse transmission *versus* peak intensity for a typical single  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  crystal. The solid line is a least-squares fit to the data points using equations 4-5 resulting in a two-photon absorption coefficient of  $8.2 \text{ GW cm}^{-1}$ . Dashed red line indicates linear loss, primarily due to scattering. b) Transmission scaled by incident intensity *versus* electric field polarization angle relative  $[100]$  crystallographic axis (offset by 8 degrees). Solid line is a least-squares fit to equation 6 and yields an anisotropy parameter of  $-0.07$ .



**Figure 3.**  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  two-photon photodetector. a) Device schematic. b) Dark current-voltage characteristic. The perovskite photoconductor has an electrical conductivity of order  $10^{-5} \Omega^{-1} \text{ cm}^{-1}$ . c) Net photocurrent (light current minus dark current) dependence on input light intensity at a bias of 50 V. The photodetector exhibits an intensity dependence which is a combination of two competing trends, the square ( $n = 2$ ) dependence of two-photon absorption, and the sublinear ( $n < 1$ ) dependence of photoconductive gain. d) Resistivity (photocurrent / input power) dependence on input intensity at a bias of 50 V. Contrary to linear photoconductors, the responsivity of the perovskite two-photon photoconductor does not decrease with increasing input intensity due to the squared dependence of absorption.



**Figure 4.** Autocorrelation of a 100 fs Ti:Sapphire laser using a  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  two-photon detector. a) Optical schematic. The input laser pulse is split into two, with one pulse variably delayed in time using a retroreflector on a motorized linear translation stage. Photocurrent generated by the two-photon perovskite photodetector is used to autocorrelate the beam pulse. b) Autocorrelation signal. We observe a 160 fs FWHM in the autocorrelation envelope, corresponding to a 100 fs pulse width. c) Electric field interference. We resolve constructive interference approximately every  $3\lambda/2$ .

1  
2  
3 ASSOCIATED CONTENT  
4

5  
6  
7 **Supporting Information Available:** Photoconductor operating under above-bandgap  
8  
9 illumination. This material is available free of charge via the Internet at <http://pubs.acs.org>.  
10  
11

12  
13  
14 AUTHOR INFORMATION15  
16  
17 **Corresponding Author**

18  
19 \* Email: ted.sargent@utoronto.ca  
20  
21

22  
23 **Author Contributions**

24  
25 The manuscript was written through contributions of all authors. All authors have given approval  
26  
27 to the final version of the manuscript. †These authors contributed equally.  
28  
29

30  
31 ACKNOWLEDGMENT

32  
33  
34 This publication is based in part on work supported by Award KUS-11-009-21, made by King  
35  
36 Abdullah University of Science and Technology (KAUST), by the Ontario Research Fund -  
37  
38 Research Excellence Program, and by the Natural Sciences and Engineering Research Council  
39  
40 (NSERC) of Canada.  
41  
42  
43

44  
45 REFERENCES

- 46  
47 (1) Xing, G.; Mathews, N.; Lim, S. S.; Yantara, N.; Liu, X.; Sabba, D.; Grätzel, M.;  
48 Mhaisalkar, S.; Sum, T. C. Low-Temperature Solution-Processed Wavelength-Tunable  
49 Perovskites for Lasing. *Nat. Mater.* **2014**, *13*, 476–480.  
50  
51 (2) Xing, G.; Mathews, N.; Sun, S.; Lim, S. S.; Lam, Y. M.; Grätzel, M.; Mhaisalkar, S.; Sum,  
52 T. C. Long-Range Balanced Electron-and Hole-Transport Lengths in Organic-Inorganic  
53 CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub>. *Science* **2013**, *342*, 344–347.  
54  
55 (3) Stoumpos, C. C.; Malliakas, C. D.; Kanatzidis, M. G. Semiconducting Tin and Lead  
56 Iodide Perovskites with Organic Cations: Phase Transitions, High Mobilities, and Near-  
57 Infrared Photoluminescent Properties. *Inorg. Chem.* **2013**, *52*, 9019–9038.  
58  
59  
60



- 1
- 2
- 3
- 4 (4) Shao, Y.; Xiao, Z.; Bi, C.; Yuan, Y.; Huang, J. Origin and Elimination of Photocurrent
- 5 Hysteresis by Fullerene Passivation in CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> Planar Heterojunction Solar Cells.
- 6 *Nat. Commun.* **2014**, *5*, 5784.
- 7 (5) Green, M. A.; Ho-Baillie, A.; Snaith, H. J. The Emergence of Perovskite Solar Cells. *Nat.*
- 8 *Photonics* **2014**, *8*, 506–514.
- 9 (6) Yang, W. S.; Noh, J. H.; Jeon, N. J.; Kim, Y. C.; Ryu, S.; Seo, J.; Seok, S. I. High-
- 10 Performance Photovoltaicperovskite Layers Fabricated Throughintramolecular Exchange.
- 11 *Science* **2015**, *348*, 1234–1237.
- 12 (7) Deschler, F.; Price, M.; Pathak, S.; Klintberg, L. E.; Jarausch, D.-D.; Higler, R.; Hüttner,
- 13 S.; Leijtens, T.; Stranks, S. D.; Snaith, H. J.; *et al.* High Photoluminescence Efficiency and
- 14 Optically Pumped Lasing in Solution-Processed Mixed Halide Perovskite
- 15 Semiconductors. *J. Phys. Chem. Lett.* **2014**, *5*, 1421–1426.
- 16 (8) Sutherland, B. R.; Hoogland, S.; Adachi, M. M.; Wong, C. T. O.; Sargent, E. H.
- 17 Conformal Organohalide Perovskites Enable Lasing on Spherical Resonators. *ACS Nano*
- 18 **2014**, *8*, 10947–10952.
- 19 (9) Tan, Z.-K.; Moghaddam, R. S.; Lai, M. L.; Docampo, P.; Higler, R.; Deschler, F.; Price,
- 20 M.; Sadhanala, A.; Pazos, L. M.; Credgington, D.; *et al.* Bright Light-Emitting Diodes
- 21 Based on Organometal Halide Perovskite. *Nat. Nanotechnol.* **2014**, *9*, 687–692.
- 22 (10) Shi, D.; Adinolfi, V.; Comin, R.; Yuan, M.; Alarousu, E.; Buin, A.; Chen, Y.; Hoogland,
- 23 S.; Rothenberger, A.; Katsiev, K.; *et al.* Low Trap-State Density and Long Carrier
- 24 Diffusion in Organolead Trihalide Perovskite Single Crystals. *Science* **2015**, *347*, 519–
- 25 522.
- 26 (11) Dong, Q.; Fang, Y.; Shao, Y.; Mulligan, P.; Qiu, J.; Cao, L.; Huang, J. Electron-Hole
- 27 Diffusion Lengths > 175 Mm in Solution-Grown CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> Single Crystals. *Science*
- 28 **2015**, *347*, 967–970.
- 29 (12) Zhu, H.; Fu, Y.; Meng, F.; Wu, X.; Gong, Z.; Ding, Q.; Gustafsson, M. V.; Trinh, M. T.;
- 30 Jin, S.; Zhu, X.-Y. Lead Halide Perovskite Nanowire Lasers with Low Lasing Thresholds
- 31 and High Quality Factors. *Nat. Mater.* **2015**, DOI: 10.1038/nmat4271.
- 32 (13) Liao, Q.; Hu, K.; Zhang, H.; Wang, X.; Yao, J.; Fu, H. Perovskite Microdisk Microlasers
- 33 Self-Assembled from Solution. *Adv. Mater.* **2015**, DOI: 10.1002/adma.201500449.
- 34 (14) Stoumpos, C. C.; Frazer, L.; Clark, D. J.; Kim, Y. S.; Rhim, S. H.; Freeman, A. J.;
- 35 Ketterson, J. B.; Jang, J. I.; Kanatzidis, M. G. Hybrid Germanium Iodide Perovskite
- 36 Semiconductors: Active Lone Pairs, Structural Distortions, Direct and Indirect Energy
- 37 Gaps, and Strong Nonlinear Optical Properties. *J. Am. Chem. Soc.* **2015**, *137*, 6804–6819.
- 38 (15) Liberman, V.; Rothschild, M.; Bakr, O. M.; Stellacci, F. Optical Limiting with Complex
- 39 Plasmonic Nanoparticles. *J. Opt.* **2010**, *12*, 065001.
- 40 (16) Chong, E. Z.; Watson, T. F.; Festy, F. Autocorrelation Measurement of Femtosecond
- 41 Laser Pulses Based on Two-Photon Absorption in GaP Photodiode. *Appl. Phys. Lett.*
- 42 **2014**, *105*, 062111.
- 43 (17) Haske, W.; Chen, V. W.; Hales, J. M.; Dong, W.; Barlow, S.; Marder, S. R.; Perry, J. W.
- 44 65 Nm Feature Sizes Using Visible Wavelength 3-D Multiphoton Lithography. *Opt.*
- 45 *Express* **2007**, *15*, 3426–3436.
- 46 (18) Zipfel, W. R.; Williams, R. M.; Webb, W. W. Nonlinear Magic: Multiphoton Microscopy
- 47 in the Biosciences. *Nat. Biotechnol.* **2003**, *21*, 1369–1377.
- 48
- 49
- 50
- 51
- 52
- 53
- 54
- 55
- 56
- 57
- 58
- 59
- 60

- 1
  - 2
  - 3
  - 4
  - 5
  - 6
  - 7
  - 8
  - 9
  - 10
  - 11
  - 12
  - 13
  - 14
  - 15
  - 16
  - 17
  - 18
  - 19
  - 20
  - 21
  - 22
  - 23
  - 24
  - 25
  - 26
  - 27
  - 28
  - 29
  - 30
  - 31
  - 32
  - 33
  - 34
  - 35
  - 36
  - 37
  - 38
  - 39
  - 40
  - 41
  - 42
  - 43
  - 44
  - 45
  - 46
  - 47
  - 48
  - 49
  - 50
  - 51
  - 52
  - 53
  - 54
  - 55
  - 56
  - 57
  - 58
  - 59
  - 60
- (19) Sheik-Bahae, M.; Said, A. A.; Wei, T.-H.; Hagan, D. J.; Van Stryland, E. W. Sensitive Measurement of Optical Nonlinearities Using a Single Beam. *Quantum Electron. IEEE J. Of* **1990**, *26*, 760–769.
- (20) Van Stryland, E. W.; Vanherzeele, H.; Woodall, M. A.; Soileau, M. J.; Smirl, A. L.; Guha, S.; Boggess, T. F. Two Photon Absorption, Nonlinear Refraction, and Optical Limiting in Semiconductors. *Opt. Eng.* **1985**, *24*, 613–623.
- (21) Van Stryland, E. W.; Woodall, M. A.; Vanherzeele, H.; Soileau, M. J. Energy Band-Gap Dependence of Two-Photon Absorption. *Opt. Lett.* **1985**, *10*, 490–492.
- (22) Sheik-Bahae, M.; Hagan, D. J.; Van Stryland, E. W. Dispersion and Band-Gap Scaling of the Electronic Kerr Effect in Solids Associated with Two-Photon Absorption. *Phys. Rev. Lett.* **1990**, *65*, 96.
- (23) DeSalvo, R.; Sheik-Bahae, M.; Said, A. A.; Hagan, D. J.; Van Stryland, E. W. Z-Scan Measurements of the Anisotropy of Nonlinear Refraction and Absorption in Crystals. *Opt. Lett.* **1993**, *18*, 194–196.
- (24) Dvorak, M. D.; Schroeder, W. A.; Andersen, D. R.; Smirl, A. L.; Wherrett, B. S. Measurement of the Anisotropy of Two-Photon Absorption Coefficients in Zincblende Semiconductors. *Quantum Electron. IEEE J. Of* **1994**, *30*, 256–268.
- (25) Hutchings, D. C.; Wherrett, B. S. Theory of Anisotropy of Two-Photon Absorption in Zinc-Blende Semiconductors. *Phys. Rev. B* **1994**, *49*, 2418.
- (26) Jishi, R. A.; Ta, O. B.; Sharif, A. A. Modeling of Lead Halide Perovskites for Photovoltaic Applications. *J. Phys. Chem. C* **2014**, *118*, 28344–28349.
- (27) Brivio, F.; Butler, K. T.; Walsh, A.; van Schilfgaarde, M. Relativistic Quasiparticle Self-Consistent Electronic Structure of Hybrid Halide Perovskite Photovoltaic Absorbers. *Phys. Rev. B* **2014**, *89*.
- (28) Giorgi, G.; Fujisawa, J.-I.; Segawa, H.; Yamashita, K. Small Photocarrier Effective Masses Featuring Ambipolar Transport in Methylammonium Lead Iodide Perovskite: A Density Functional Analysis. *J. Phys. Chem. Lett.* **2013**, *4*, 4213–4216.
- (29) Even, J.; Pedesseau, L.; Jancu, J.-M.; Katan, C. Importance of Spin-Orbit Coupling in Hybrid Organic/Inorganic Perovskites for Photovoltaic Applications. *J Phys Chem Lett* **2013**, *4*, 2999–3005.
- (30) Mosconi, E.; Amat, A.; Nazeeruddin, M. K.; Grätzel, M.; De Angelis, F. First-Principles Modeling of Mixed Halide Organometal Perovskites for Photovoltaic Applications. *J. Phys. Chem. C* **2013**, *117*, 13902–13913.
- (31) Pedesseau, L.; Jancu, J.-M.; Rolland, A.; Deleporte, E.; Katan, C.; Even, J. Electronic Properties of 2D and 3D Hybrid Organic/inorganic Perovskites for Optoelectronic and Photovoltaic Applications. *Opt. Quantum Electron.* **2014**, *46*, 1225–1232.
- (32) Hutchings, D. C.; Wherrett, B. S. Theory of the Polarization Dependence of Two-Photon Absorption in Zinc-Blende Semiconductors. *J. Mod. Opt.* **1994**, *41*, 1141–1149.
- (33) Konstantatos, G.; Clifford, J.; Levina, L.; Sargent, E. H. Sensitive Solution-Processed Visible-Wavelength Photodetectors. *Nat. Photonics* **2007**, *1*, 531–534.
- (34) Hu, X.; Zhang, X.; Liang, L.; Bao, J.; Li, S.; Yang, W.; Xie, Y. High-Performance Flexible Broadband Photodetector Based on Organolead Halide Perovskite. *Adv. Funct. Mater.* **2014**, *24*, 7373–7380.
- (35) Miller, A.; Reid, D. T. *Ultrafast Photonics*; Scottish Graduate Series; CRC Press, 2004.