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Achieving Uniform Carriers Distribution in MBE Grown Compositionally Graded InGaN Multiple-Quantum-Well LEDs

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Abstract: We investigated the design and growth of compositionally-graded InGaN multiple quantum wells (MQW) based light-emitting diode (LED) without an electron-blocking layer (EBL). Numerical investigation showed uniform carrier distribution in the active region, and higher radiative recombination rate for the optimized graded-MQW design, i.e. $\text{In}_{0-x}\text{Ga}_{1-(1-x)}\text{N} / \text{In}_x\text{Ga}_{(1-x)}\text{N} / \text{In}_{x-0}\text{Ga}_{(1-x)-1}\text{N}$, as compared to the conventional stepped-MQW-LED. The composition-grading schemes, such as linear, parabolic, and Fermi-function profiles were numerically investigated for comparison. The stepped- and graded-MQW-LED were then grown using plasma assisted molecular beam epitaxy (PAMBE) through surface-stoichiometry optimization based on reflection high-energy electron-diffraction (RHEED) *in-situ* observations. Stepped- and graded-MQW-LED showed efficiency roll over at 160 A/cm^2 and 275 A/cm^2 , respectively. The extended threshold current density roll-over (droop) in graded-MQW-LED is due to the improvement in carrier uniformity and radiative recombination rate, consistent with the numerical simulation.

Index terms: Compositional grading, Light emitting diodes (LEDs), polarization field, semiconductor quantum well, solid state lighting, wavefunction overlap.

1. Introduction

Despite the InGaN based LEDs being the key driver for solid-state lighting technology, the device performance at high injection current density is limited by efficiency droop. A number of possible mechanisms had been suggested to account for the droop, summarized as follows: Auger recombination [1], carrier delocalization [2], current injection efficiency quenching [3]. The severe band bending in c-plane InGaN/GaN multiple quantum well (MQW) arising from the built-in spontaneous and piezoelectric polarization fields [4-6] was also accounted for the major root cause of the issue. As the severe band bending leads to the reduced oscillator strength, which is inversely proportional to the spatial electron-hole separation, non-uniform distribution of carriers in MQW, and electron leakage, especially at the QW closest to p-GaN [7-9].

In typical LED structures, the electron blocking layer (EBL) has been incorporated between the active region and p-GaN to increase overall carrier concentration by reflecting electrons back into the active region. The downside of the EBL scheme is the reduction of the hole injection efficiency, due to severe band bending and valance band (VB) offset at AlGaIn/GaN interface which can promote electron leakage [10-12]. Solutions suggested to partly mitigate the side-effects of EBL are: graded-EBL [13], lattice-matched InAlN-EBL [14, 15], superlattice-EBL [16, 17] and, graded-superlattice-EBL [8, 18]. EBL-less designs based on insertion of undoped-GaN-layer in between active-region and p-GaN layer [19], GaN-AlGaIn-GaN as last barrier [20], AlGaIn-step-like-barriers [21], thin-AlGaIn-barriers [22], specially-designed p-InGaIn barrier [23], and two-step Mg-doped p-GaN [24] were also proposed to improve carrier confinement. Although the carrier concentration improves in these theoretical and experimental studies, the issues of non-uniform carrier distribution remain, which may again lead to efficiency droop of MQW. Thus, active region design that promotes uniform carriers distribution for reducing electron leakage, and improving hole injection become important. One

of the possible ways to increase the output power and alleviate efficiency droop, is to improve the wavefunction overlap in the InGaN QWs by using QWs with non-conventional shapes [25-31], semi-polar [32, 33] or non-polar [34, 35] QWs as active material in LED structures. InGaN/GaN graded MQW LED with EBL based on linear grading realized using MOCVD showed ~ 2.5 times larger quantum efficiency rollover threshold as compared to the conventional stepped-MQW-LED [29]. This improvement has been attributed to the reduced polarization field, and reduced band bending in the graded-MQW based active region.

In this study, we design graded-MQW active region to achieve uniform carrier distribution with high radiative recombination in the active region, without resorting to AlGaIn-based EBL or QW-barrier large bandgap insertion layers. In agreement with the obtained simulation results, extended quantum efficiency rollover threshold was obtained for graded-MQW-LED as compared to the conventional stepped-MQW-LED. To investigate the effect of grading profile on carrier distribution and recombination rate considering growth implementation, we have studied various graded schemes, *i.e.* the linear, parabolic and Fermi-function profiles in the simulation. A graded-MQW-LED was grown on c-plane sapphire template substrate using plasma assisted molecular beam epitaxy (PAMBE), and the efficiency droop was compared with a stepped-MQW-LED. Micro-LEDs with circular mesa diameter (D) of 80 μm were fabricated.

2. Design and Simulation Results

Figure 1 shows the designed layer structures for the stepped- and graded-MQW LEDs on sapphire-based GaN template substrate. The LED structures consist of 650 nm thick n-GaN layer, followed by five pairs of QWs and 100 nm p-GaN cap layer. The active regions of the simulated stepped-MQW-LED consist of five pairs of In_{0.2}Ga_{0.8}N (3nm) / GaN (12nm), while that of graded-MQW consists of GaN (12nm) / In_{0.2}Ga_{1-0.8}N (1.75nm) / In_{0.2}Ga_{0.8}N (0.5nm) / In_{0.2-0}Ga_{0.8-1}N (1.75nm) / GaN (12nm) QWs respectively. Simulations were performed considering energy band alignment, comparison of carrier densities in MQW, spatial wavefunction overlap, electric field, and recombination rates using the NEXTNANO software [36]. The active region of simulated LED structure was assigned an unintentional background electron carrier concentration of 1x10¹⁶ cm⁻³ to prevent doping induced non-radiative recombination, and impurity scattering. The conduction-to-valance-band offsets ratio, ΔE_c/ΔE_v, was taken as 70:30 for the material system [37]. Band parameters for ternary alloy InGaIn were calculated based on *Vegard's* law [38]. The electron and hole, effective masses, and the elastic, piezoelectric and deformation potentials for the ternary compound were extrapolated linearly [39]. The *Arora* model was used to obtain doping dependent mobilities for the intrinsic InGaIn [40]:

$$u(T, N_i) = u_{min}^{n,p} \left(\frac{T}{T_o}\right)^{\alpha_m^{n,p}} + \frac{u_d^{n,p} \left(\frac{T}{T_o}\right)^{\alpha_d^{n,p}}}{1 + \left(\frac{N_D + N_A}{N_0^{n,p} \left(\frac{T}{T_o}\right)^{\alpha_N^{n,p}}}\right)^{A_a^{n,p} \left(\frac{T}{T_o}\right)^{\alpha_a^{n,p}}}}$$

where $u_{min}, u_d, \alpha_m, \alpha_d, \alpha_N, \alpha_a, N_0$ are Arora model fitting parameters [41]. $N_A + N_D$ is the total concentration of ionized impurities.

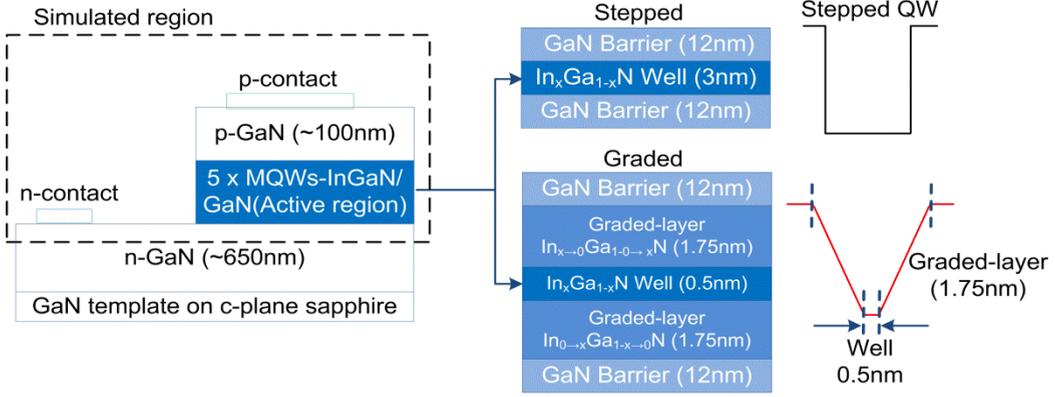


Figure 1: Schematic diagram of the InGaN LED structures with-stepped-MQW, and the compositionally graded-MQW.

Figure 2(a) shows the calculated energy band diagram with the electrons and holes quasi-Fermi-levels (E_{fn} and E_{fp}) for the stepped-MQW-LED at a forward bias of 3 V and at room temperature. Abruptness at the polarization mismatch hetero-interfaces (GaN/InGaN stepped MQW) leads to large sheet charges at the interfaces, which result in triangular shaped QW as shown in figure 2(a). For simplicity, five stacks of QW along the [0001] growth direction are indicated by QW1-5 with QW1 adjacent to the n-GaN and QW5 adjacent to p-GaN. The E_{fp} in the stepped-MQW-LED mimics a staircase pattern with effective potential barriers of 848 meV, 848 meV, 821 meV, 713 meV and 622 meV in QW1-5 respectively. This leads to the non-uniform distribution of hole concentrations with maximum and minimum values in QW5 and QW1, correspondingly (see figure 2(b)). Non-uniform electron concentration among the five QWs was also obtained due to overall downward bending of conduction band (CB) along the growth direction. Thus, most electrons get accumulated in the QW5. Such non-uniform distribution of carrier concentrations among the QWs in the active region will be detrimental for device performance.

A graded-MQW scheme was conceived and optimized in this work. Figure 2(c) shows the energy band diagram of a graded-MQW-LED in which the 3 nm thick stepped QW was replaced by 1.75 nm linearly down-graded indium composition layer, 0.5-nm-thick $\text{In}_{0.2}\text{Ga}_{0.8}\text{N}$ and 1.75 nm linearly up-graded indium composition layer, which resulted in a reduced polarization mismatch. This leads to the absence of staircase E_{fp} profile. Moreover, since the active region and p-GaN see equal potential barrier of 492 meV, the uniform distribution of carriers are greatly facilitated into the five quantum wells in the linearly graded-MQW-LED as shown in figure 2(d). In addition, improved spatial overlap for graded-MQW (for QW3 19.6%) as compared to stepped-MQW (for QW3 4.9%) was obtained. To shed light on the role of a thin compositionally graded layer in reducing the abruptness of polarization mismatch in achieving uniform carrier distribution in the active region, the electrical field profile was examined. As evident in figure 3, we obtained lower electric fields in the active region of the graded-MQW-LED as compared to that of stepped-MQW-LED, resulting in a lower band

bending and hence a more uniform carrier distributions within the active region of graded-MQW-LED.

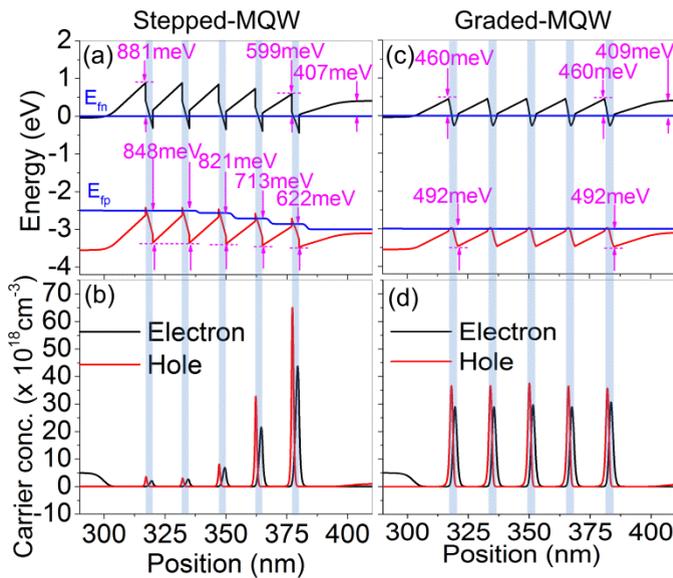


Figure 2: The respective energy band diagram and carrier concentration at 3V bias voltage for stepped-MQW in (a) and (b) and for graded MQW-LED (c) and (d), respectively (shaded regions represent the QW; QW1-5 are positioned from left to right).

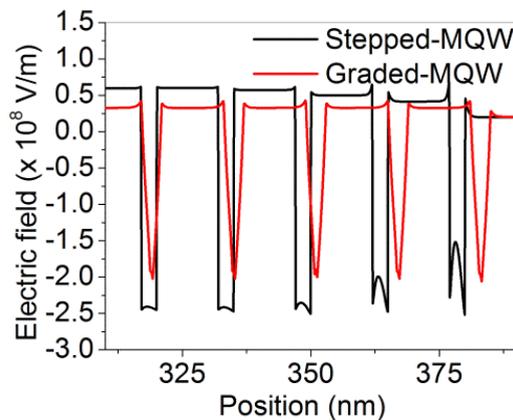


Figure 3: Calculated electric field of stepped-MQW-LED and graded-MQW-LED under the bias of 3V.

In general, compositional grading in InGaN graded-layers is experimentally implemented by employing change in growth temperature [26, 28-31]. However, the resulting grading in composition may not be perfectly linear [42-44]. Also, the role of non-linear grading profiles on carrier distribution in MQW based LEDs remain unaddressed [45, 46]. Thus we further investigated the effect of different grading profile, such as linear, parabolic, and Fermi-function numerically, on the carrier distribution and radiative recombination for the graded-MQW-LED design compared to the stepped-MQW-LED.

As shown in figure 4(a), all three grading schemes resulted in comparable radiative recombination across all five QWs while stepped-MQW-LED showed disparate radiative

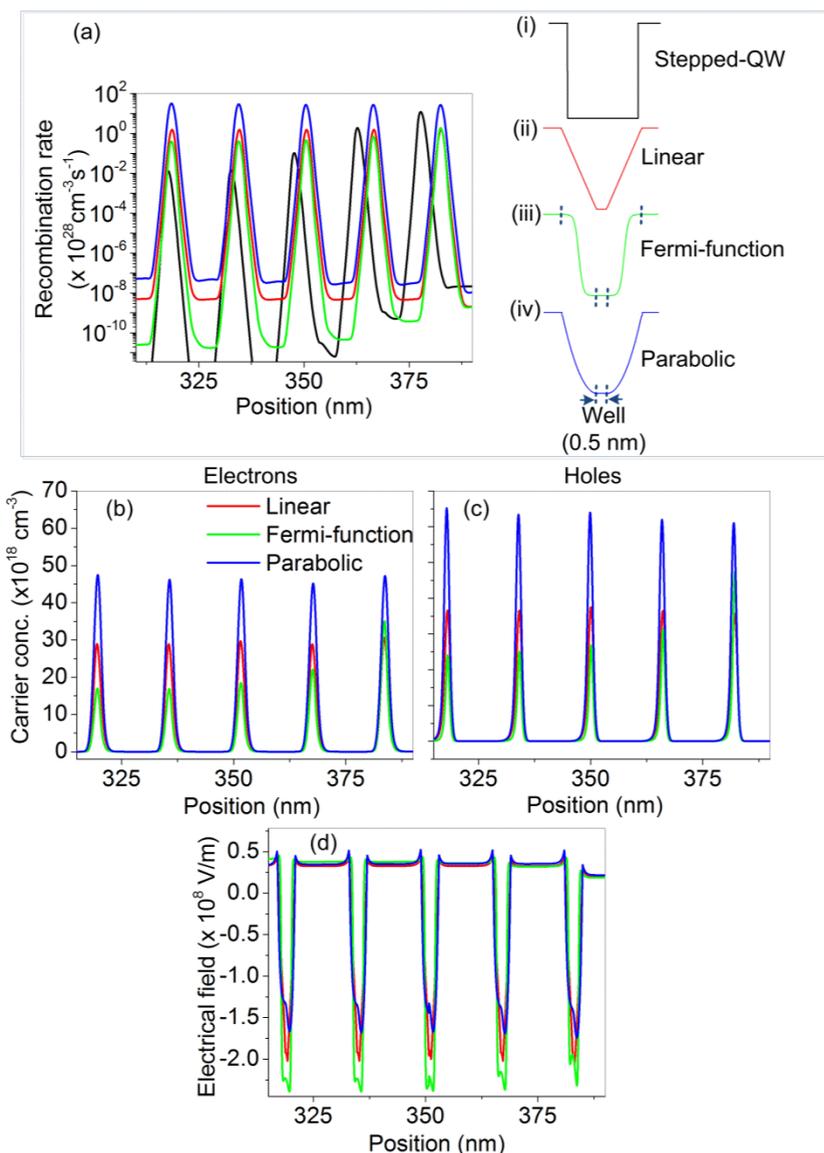


Figure 4: (a) Radiative recombination rate of compositionally graded-MQW-LEDs with different indium grading schemes (linear, Fermi-function, parabolic) and stepped-MQW-LED (schematics of bandgap profile for different schemes were also shown). Electrons (b) and, hole (c) concentrations in graded-MQW-LEDs with different compositional grading profile (linear, Fermi-function, and parabolic). (d) The calculated electric field of graded-MQW-LED with the linear (red curve), Fermi-function (green curve), and parabolic (blue curve) grading profiles under the bias of 3V.

recombination as large as three orders of comparing between QW1 and QW5. Moreover, graded-MQW-LED with parabolic compositional profile attained the highest radiative recombination rate of $3.3 \times 10^{29} \text{ cm}^{-3} \text{ s}^{-1}$ which is attributed to its higher electron and hole concentrations obtained in QW1-5 as compared to linear and Fermi-function profiles (see figure 4(b) and (c)). The simulated results showed that the fundamental transition in all the QWs (QW1-5), with parabolic profile, was at 2.89eV with 3V forward bias. In order to verify the highest radiative rate with parabolic profile, we further examined electric field in the active regions with different grading profiles. As shown in figure 4(d), the lowest electrical field was obtained using parabolic profile based graded-MQW, and as a consequence, lower band

bending and higher electron-hole wavefunction overlap was obtained. Thus, the parabolic shape has a high radiative rate as a result of high carrier concentration and high electron-hole wavefunction overlap.

3. Device Implementation and Measurements

For experimental verification of the advantageous effect of the insertion of compositionally graded layers in graded-MQW-LED, it is important to consider suitable growth technology for implementing graded-MQW-LED. Comparing with MOCVD, PAMBE growth process is advantageous for the growth of thin (few monolayers) compositionally graded InGaN layer since it is capable of low growth rate process, and equipped with *in-situ* reflection high-energy electron diffraction (RHEED) for monitoring surface growth condition at monolayer scale.

Thus, we have implemented a graded-MQW-LED by using PAMBE, consisting of 650 nm thick n-GaN layer, followed by five pairs of graded-QWs, and 100 nm p-GaN cap layer. In the graded-QW region, the undoped 12 nm GaN quantum barriers were grown under stoichiometric conditions using Ga beam equivalent pressure (BEP) of 6.0×10^{-8} Torr at 595 °C. The indium flux 8.5×10^{-8} Torr and nitrogen plasma conditions, i.e. flow rate 0.4 sccm and RF power 200 W were kept constant during the well and graded-layer growth. An interruption time of 1 min was provided for ramping the growth temperature from 595 to 585 °C, and vice-versa, before the growth of 1.75 nm graded-InGaN cladded 0.5 nm InGaN QW. The interruption time was optimized in separate growth experiments utilizing RHEED technique by measuring the specular spot intensity for ensuring complete desorption of excess gallium before the growth of graded- and well-layers. The growth kinetics for graded-MQW were also optimized to fulfil ($f_{Ga} + f_{In}$) > f_N , ($f_{Ga} < f_N$) conditions, under which indium composition, x , in the InGaN alloy is governed by following expression [47, 48];

$$x = \frac{f_N - f_{Ga}}{f_N}$$

where, f_{Ga} , f_{In} , f_N are the incident fluxes of Ga, In and N. According to the above equation, the required indium composition can be varied by changing the f_{Ga} at a given growth temperature. Therefore, to realize graded indium profile at the bottom, and top graded-layers, we changed the gallium BEP between 6.0×10^{-8} Torr and 4.5×10^{-8} Torr by varying the effusion cell temperature linearly. It is expected that a parabolic compositional grading is achieved with linear variation in gallium effusion cell temperature.

Comparing growth aspects, our EBL-less design based on graded-MQW-LED is simpler to implement as compared to the previously proposed EBL-less design based on AlGaIn-in-active-region [20-22]. In order to avoid dissociation of InGaN quantum well, AlGaIn layers need to be grown at lower growth temperature (around growth temperature of InGaN). At such temperature, grown AlGaIn leads to poor crystal quality, in particular, high defect density. However, our EBL-less design overcomes such ambiguity in the growth process. Figure 5(a) shows electroluminescence spectra of graded-MQW-LED having peak emission wavelength of 450 nm with full-width-half-maximum (FWHM) of 27.85 nm at injection current density 350 A/cm². Also, side hump is observed at around 430 nm marked with a dotted circle, which point

towards the availability of lower Indium compositional region within the well i.e. incorporated graded-layer. Figure 5(b) shows the injection current density vs. the DC bias characteristics for graded-MQW-LED with current density of about $\sim 400 \text{ A/cm}^2$ at 7.5 V. Figure 5(b) inset shows the band diagram of the graded-QW with parabolic profile along with its corresponding electron and hole wavefunction having spatial overlap ($\Gamma_{e,hh}$) of 19%.

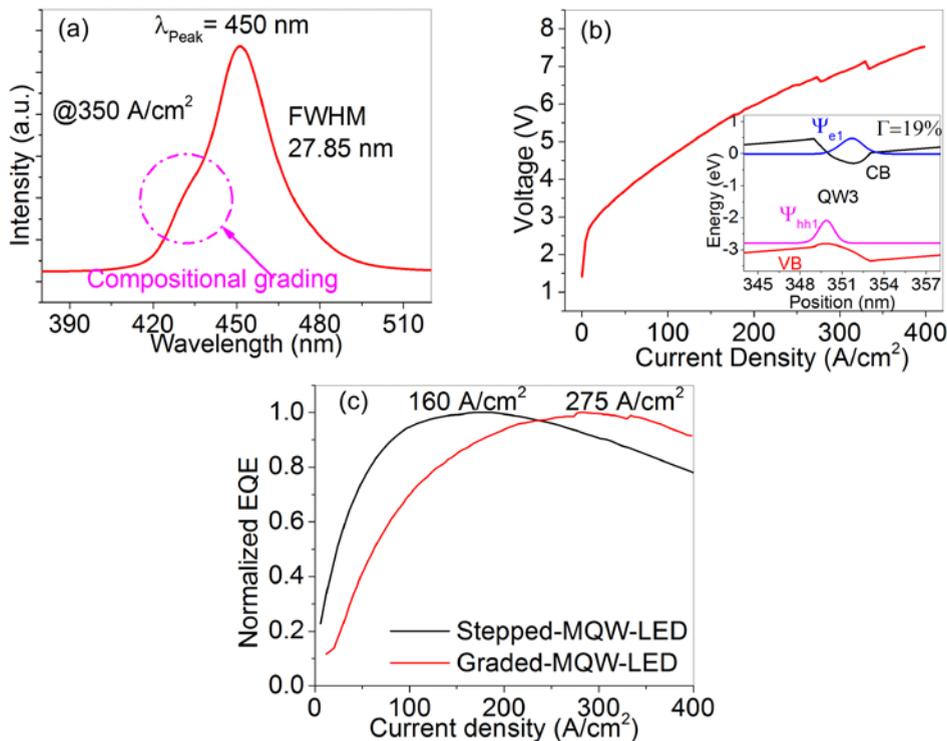


Figure 5: (a) EL spectra of graded-MQW-LED at 350 A/cm^2 (b) Typical J-V characteristics of graded-MQW-LED (inset shows energy band diagram of graded-QW3 with parabolic profile at bias voltage of 3V). (c) Normalized EQE of graded MQW-LED and stepped-MQW-LED.

Further, external quantum efficiencies (EQEs) for graded-MQW-LED and stepped-MQW-LED were also evaluated by measuring the light output powers under different injection current densities. The advantage of the graded-MQW-LED is evident in the normalized external quantum efficiency plot in figure 5(c). The stepped-MQW-LED attained maximum efficiency at injection level of 160 A/cm^2 . On the other hand, the maximum efficiency of graded-MQW-LED was attained at injection level of 275 A/cm^2 . Hence, the onset of efficiency rollover for graded-MQW-LED occurred at higher injection current level as compared to the stepped-MQW-LED. The extended droop roll-over for graded-MQW-LED can be attributed to the reduced band bending, uniform carrier distribution in the active region and non-overcrowding of carriers in the last-QW closest to the p-GaN layer.

As the carriers are evenly distributed across all QWs, the last graded-QW has a lower carrier concentration compared to that of the stepped-QW, resulting in the alleviation of the undesired carrier leakage and Auger recombination, and thus contributed to the extended efficiency roll-over effect. The evenly distributed carriers in all QWs achieved mainly due to the reduced polarization field by implementing compositionally graded-layer, and improved hole injection into the active region by using EBL-less design.

4. Conclusions

In conclusion, we have compared LED designs based on graded-MQW and conventional stepped-MQW; the graded-MQW having enhanced carrier uniformity due to reduced polarization field as obtained numerically. Different grading schemes based on linear, parabolic and Fermi-function were taken into account. In particular, graded-MQW-LED with parabolic grading profile showed the highest radiative recombination rate with uniform carrier distribution. Our work highlights the realization of optimized graded-MQW-LED by surface stoichiometry monitoring using RHEED based on the controllable growth rate in PAMBE. The efficiency droop in the implemented graded-MQW-LED occurred at higher injection levels ($\sim 275 \text{ A/cm}^2$) as compared to a stepped-MQW-LEDs ($\sim 160 \text{ A/cm}^2$), thus, demonstrating the effectiveness of our EBL-less graded active region design.

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